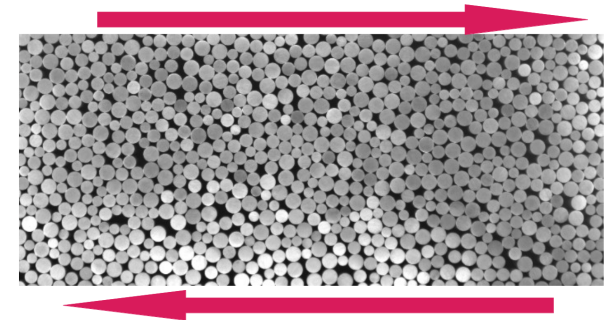


# The fate of shear-oscillated amorphous solids

C. Liu, EE Ferrero, EA Jagla, K Martens, A Rosso, L Talon, *J. Chem. Phys.* 156, 104902 (2022)

Ezequiel Ferrero

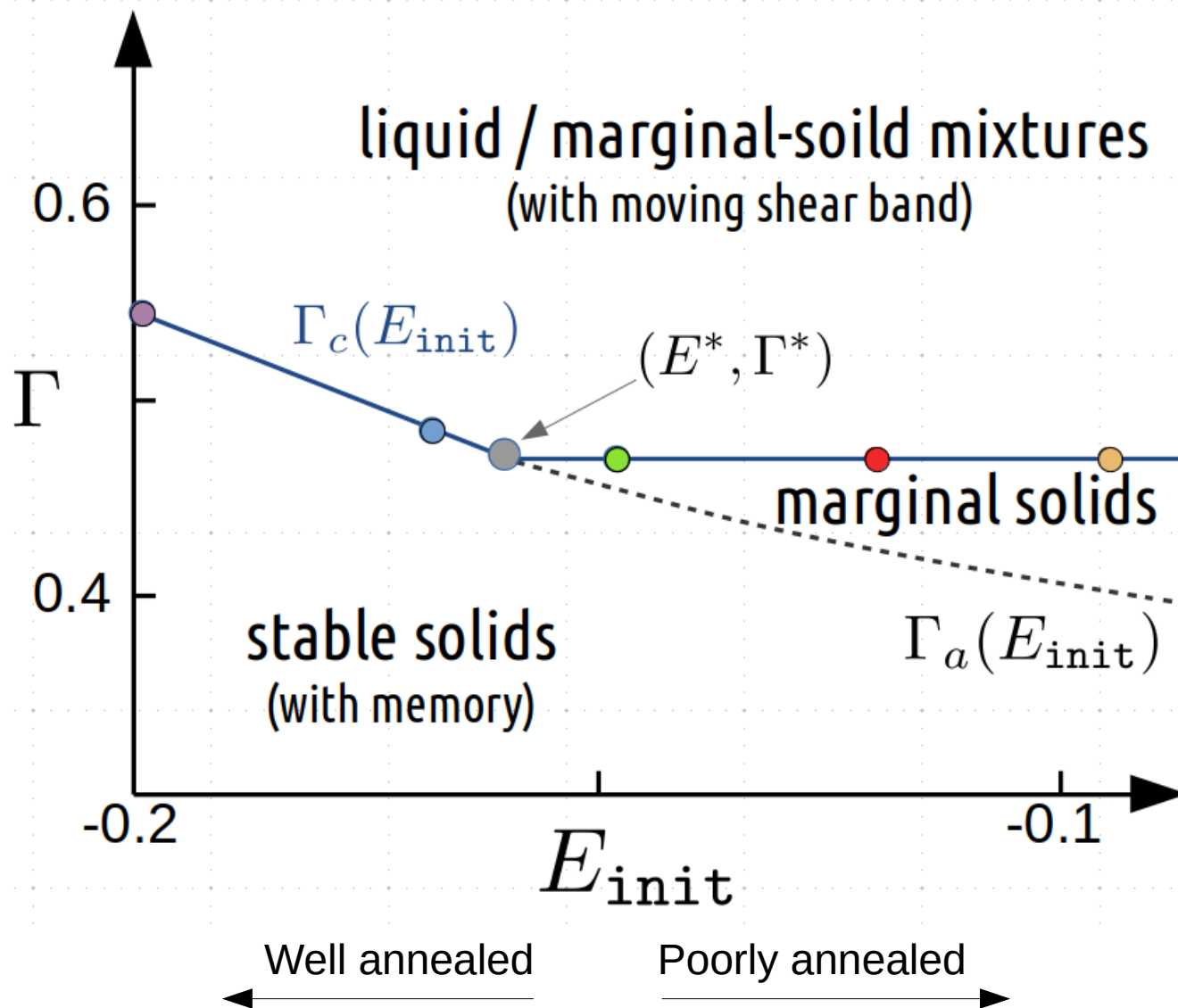


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Condensed Matter Theory Department, Universitat de **Barcelona** (Spain)

IDE-GDR workshop – Grenoble (France)  
November 30th 2022

# The fate of shear-oscillated amorphous solids

Steady-state is fully determined by the initial preparation and the oscillation amplitude

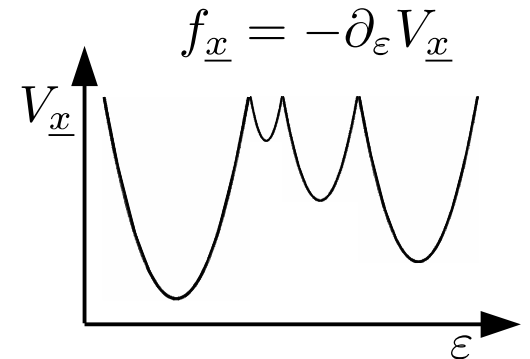


# Scalar 2D model: allowing plasticity on a strain component

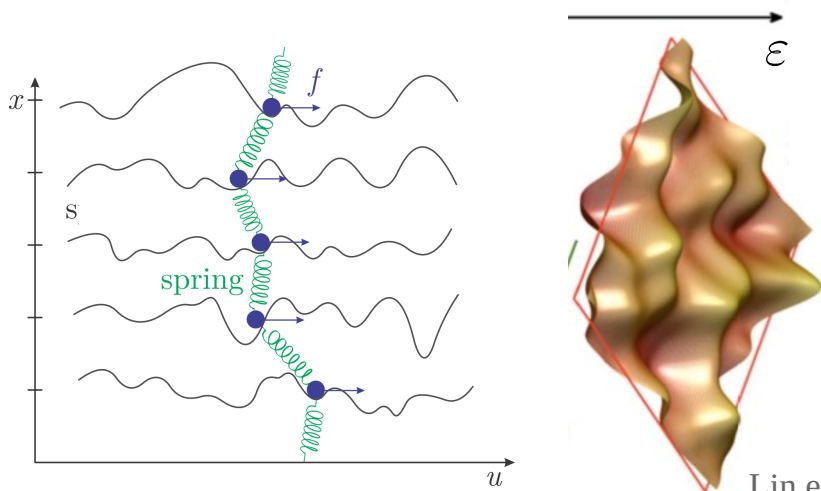
$$E_{\text{elast.}} = \int d^2 \mathbf{r} (B \varepsilon_1^2 + \mu \varepsilon_2^2 + \mu \varepsilon_3^2) \longrightarrow E_{\text{plast.}} = \int d^2 \mathbf{r} (B \varepsilon_1^2 + \mu \varepsilon_2^2 + V_x[\mathbf{r}, \varepsilon_3])$$

E. A. Jagla, PRE 76, 046119 (2007), PRE 101, 043004 (2020), X. Cao et al., Soft Matter 14, 3640 (2018)

$$\partial_t \varepsilon(\mathbf{r}, t) = \underbrace{\int d^2 \mathbf{r}' G(\mathbf{r} - \mathbf{r}') \varepsilon(\mathbf{r}')}_{\text{"Eshelby" interaction}} + \underbrace{f_x(\mathbf{r}, \varepsilon)}_{\text{pinning force}} + \underbrace{\Sigma_{\text{ext}}}_{\text{applied stress}}$$



Each site w/ independent disordered potential



Lin et al. PNAS '14

**By construction: dynamical disorder, no memory**  
 once a local strain overcomes a barrier of the parabolic basin, **the basin changes** irreversibly

**Not quenched!**

To control the macroscopic strain we use

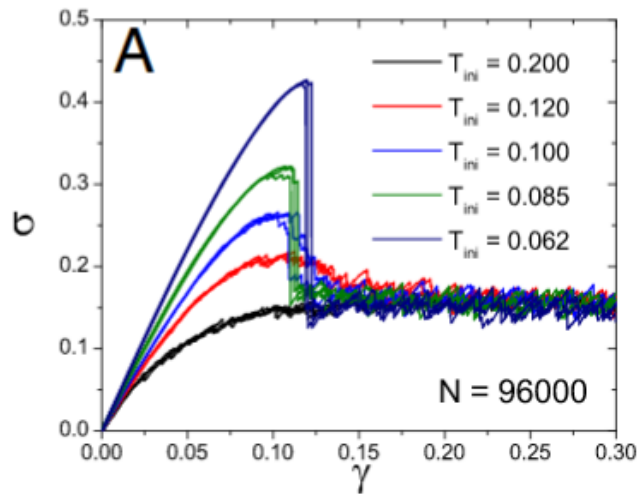
$$\Sigma_{\text{ext}} \equiv \kappa(\gamma - \bar{\varepsilon})$$

Like the "elastic line", but in 2D w/Eshelby-like interactions

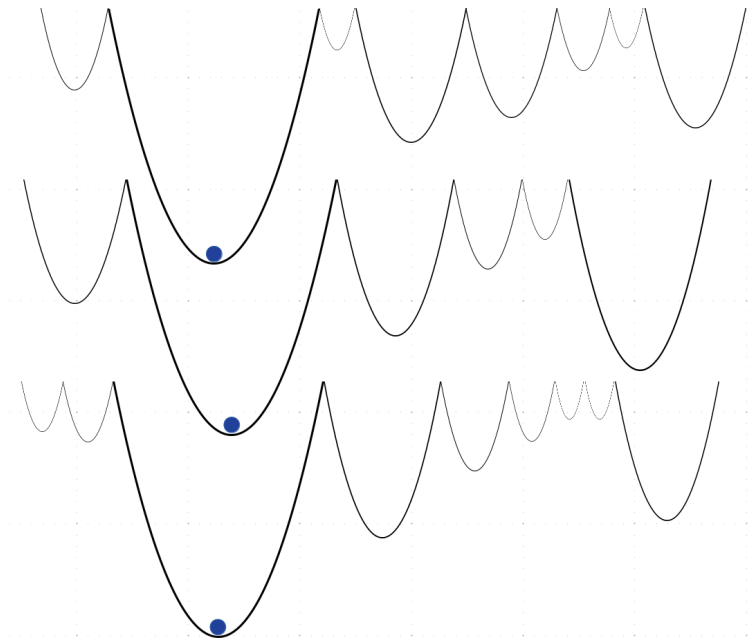
# The initial configuration

From MD we know: stress overshoot, yield stress and strain **depends on sample preparation**

We ‘**mimic**’ different **degrees of annealing** by starting from a highly-sheared sample and transforming its sitting wells



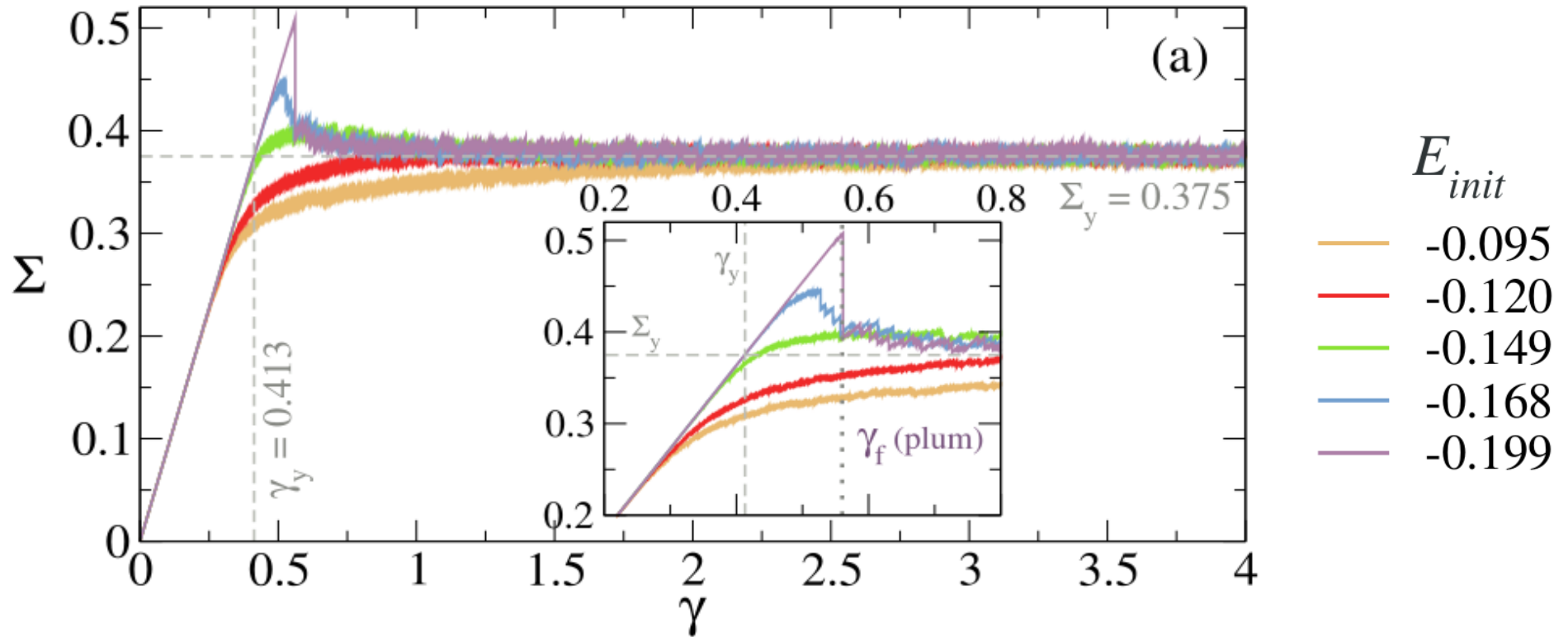
Ozawa et al. PNAS **115**, 6656 (2018)



“Ductile yielding” vs. “Fragile yielding”

$E_{init}$  decreases as we make deeper and align the wells

# Uniform (quasistatic) deformation: steady liquid



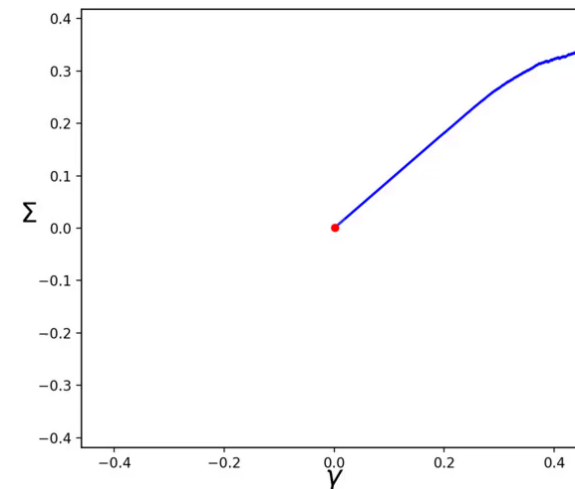
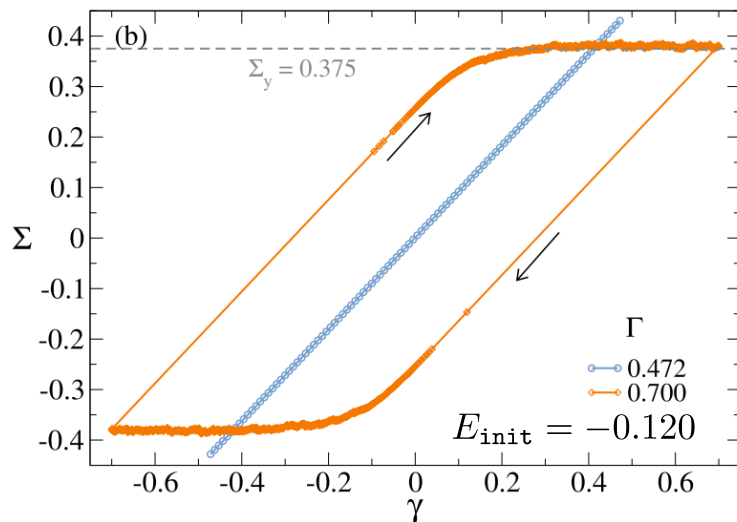
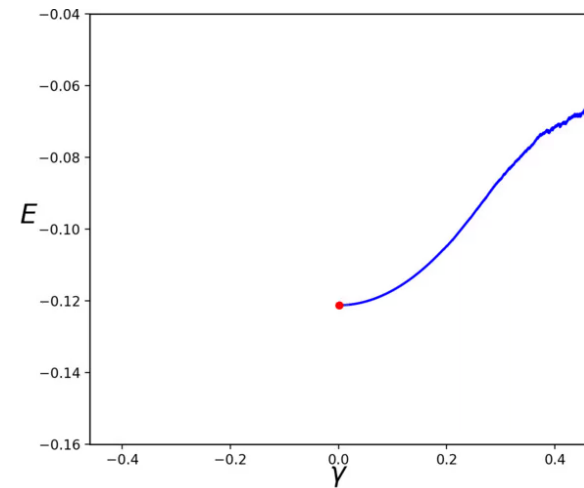
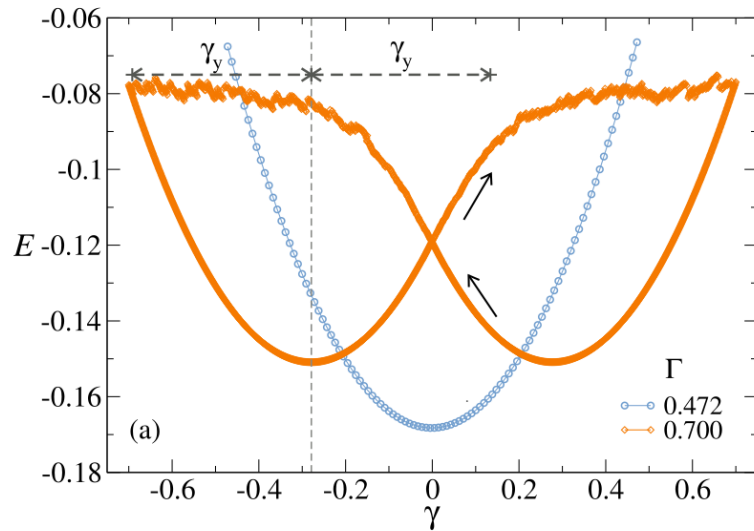
Plasticity becomes important at

$$\gamma_y = \Sigma_y / \mathcal{G} \simeq 0.413$$

# Oscillatory deformation: 2 steady phases

Starting at  $\gamma=0$  we deform (quasistatically) up to  $+\Gamma$ , then to  $-\Gamma$ , then back 0, and so on...

**Two steady-states:** solid phase ( $\Gamma < \Gamma_c$ ), mixed phase ( $\Gamma > \Gamma_c$ )

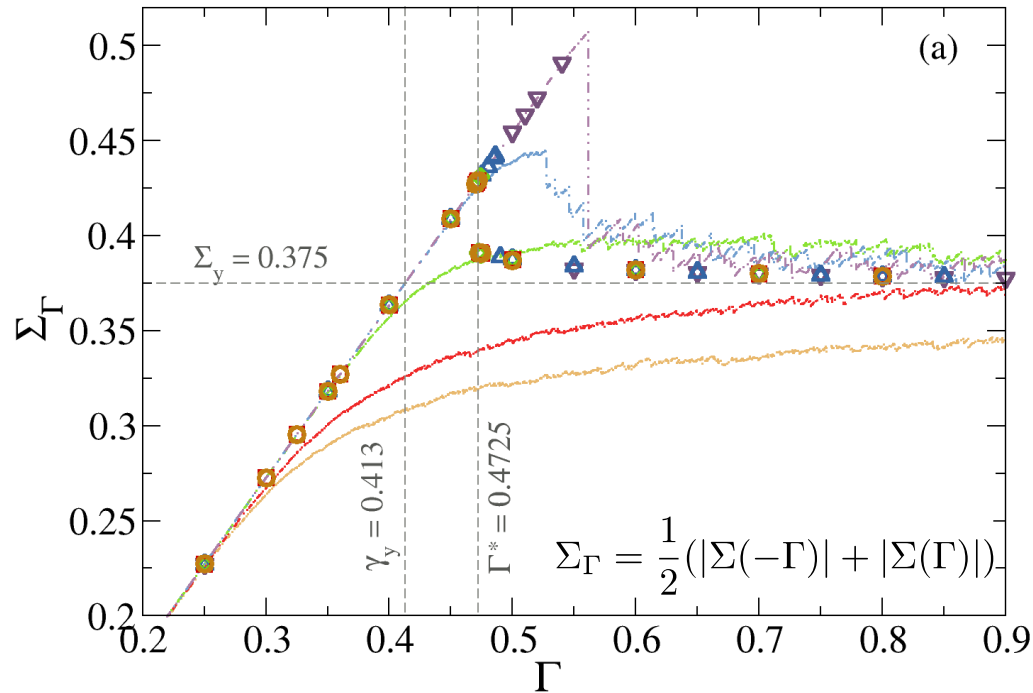


Steady state

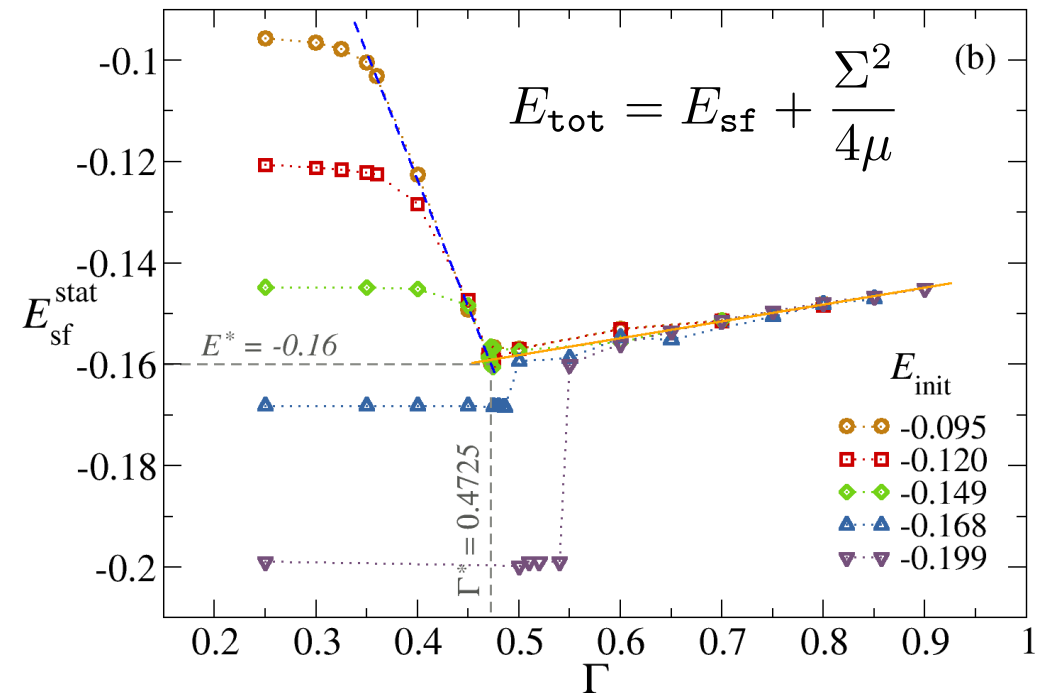
Transient evolution  $\Gamma = 0.45 < \Gamma_c$

# Steady stress and energy show sharp jumps

Stress at maximum strain



Per-site “stress-free” energy



Sharp stress jump at a critical amplitude  $\Gamma_c(E_{\text{init}})$

If  $(E_{\text{init}} > E^*) \longrightarrow \Gamma_c(E_{\text{init}}) = \Gamma^*$

“critical annealing level”

Above  $\Gamma_c(E_{\text{init}})$ , the stress of all samples is indistinguishable from  $\Sigma_y$  (uniform deformation)

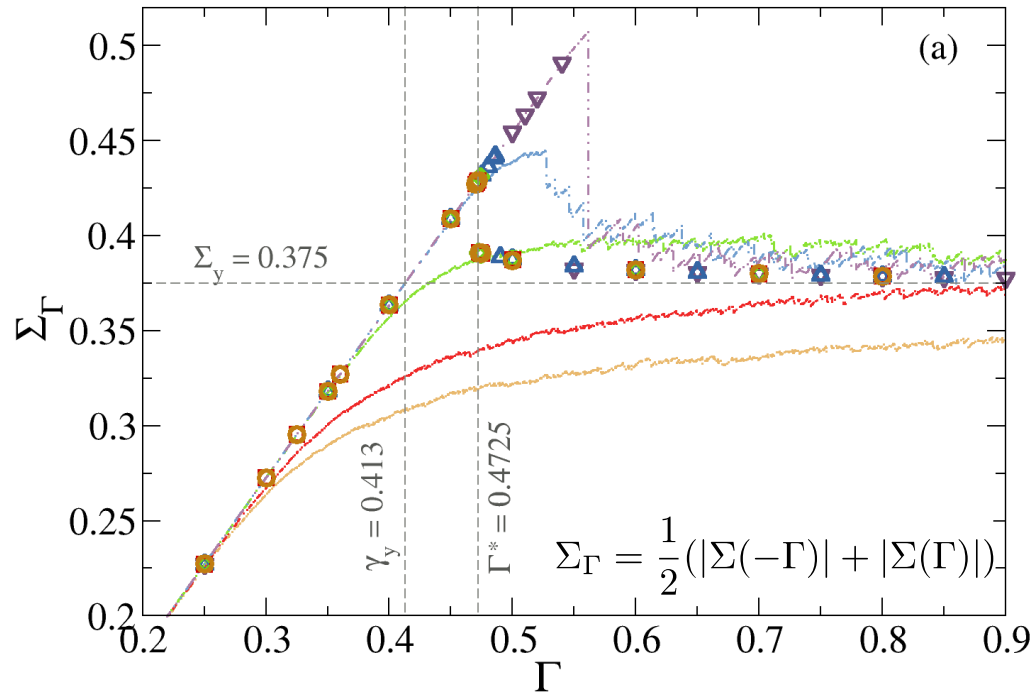
For very small  $\Gamma$ ,  $E_{\text{sf}}$  strongly depends on  $E_{\text{init}}$

Increasing  $\Gamma$  within the solid phase ( $\Gamma < \Gamma_c$ ):

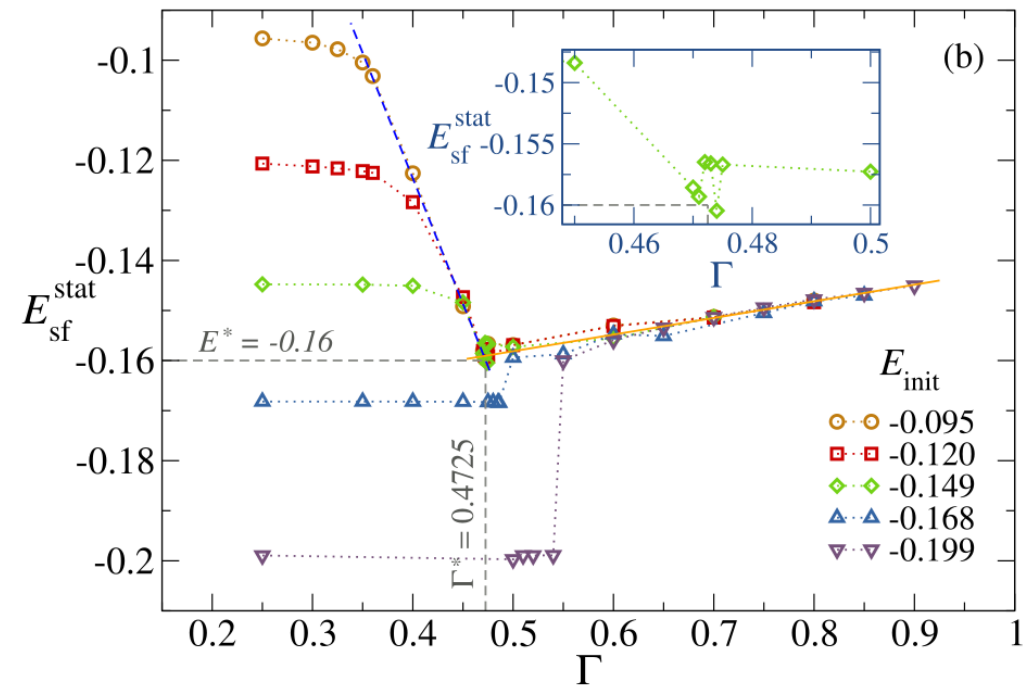
- Samples with  $E_{\text{init}} < E^* = -0.16$  **keep memory**
- Samples with  $E_{\text{init}} > E^*$  **shear anneal**

# Steady stress and energy show sharp jumps

Stress at maximum strain



Per-site “stress-free” energy



Sharp stress jump at a critical amplitude  $\Gamma_c(E_{init})$

If  $(E_{init} > E^*) \longrightarrow \Gamma_c(E_{init}) = \Gamma^*$

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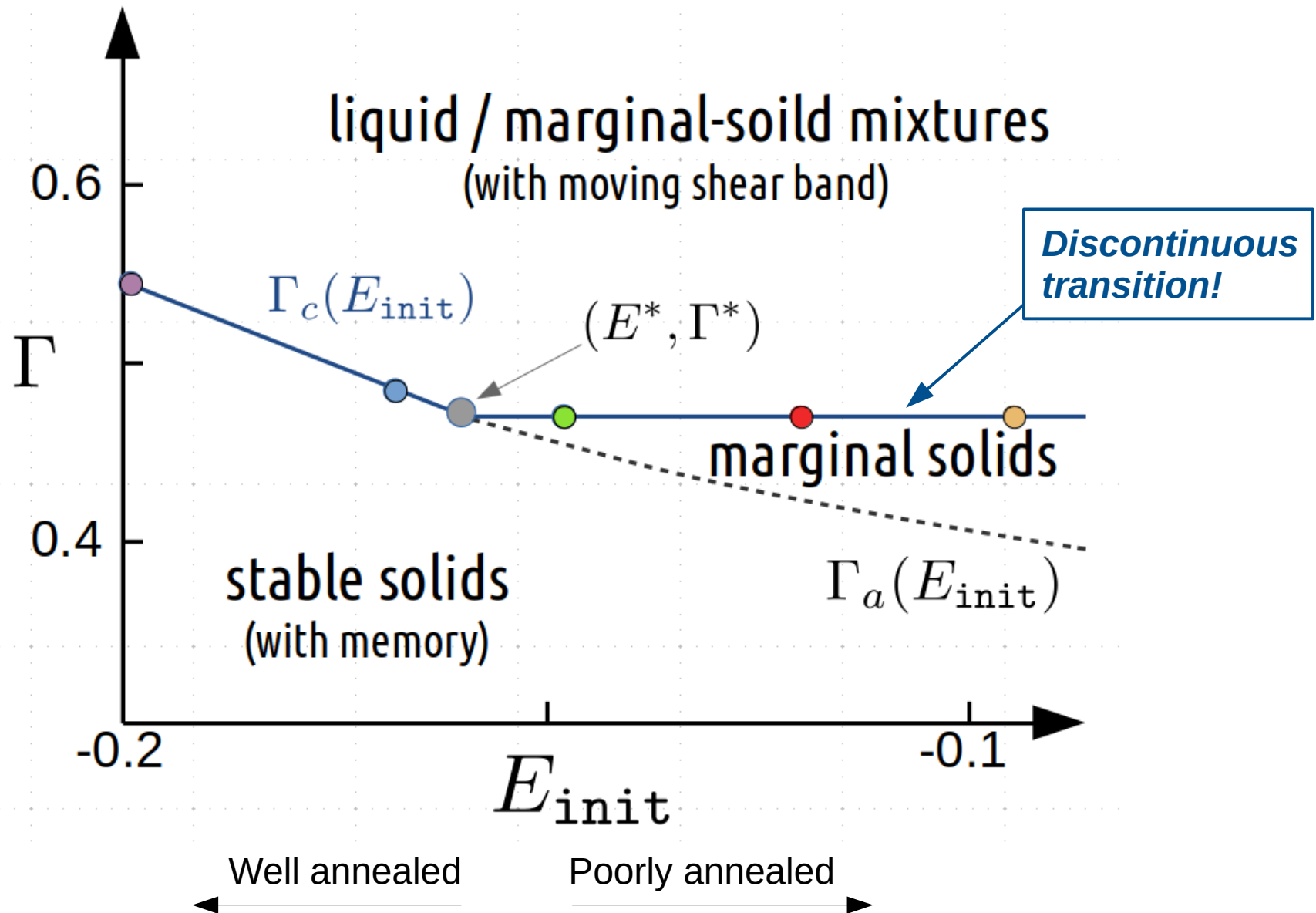
Above  $\Gamma_c(E_{init})$ , the stress of all samples is indistinguishable from  $\Sigma_y$  (uniform deformation)

Even when it’s less evident, the **energy  $E_{sf}$**  also has a **discrete jump** at the transition

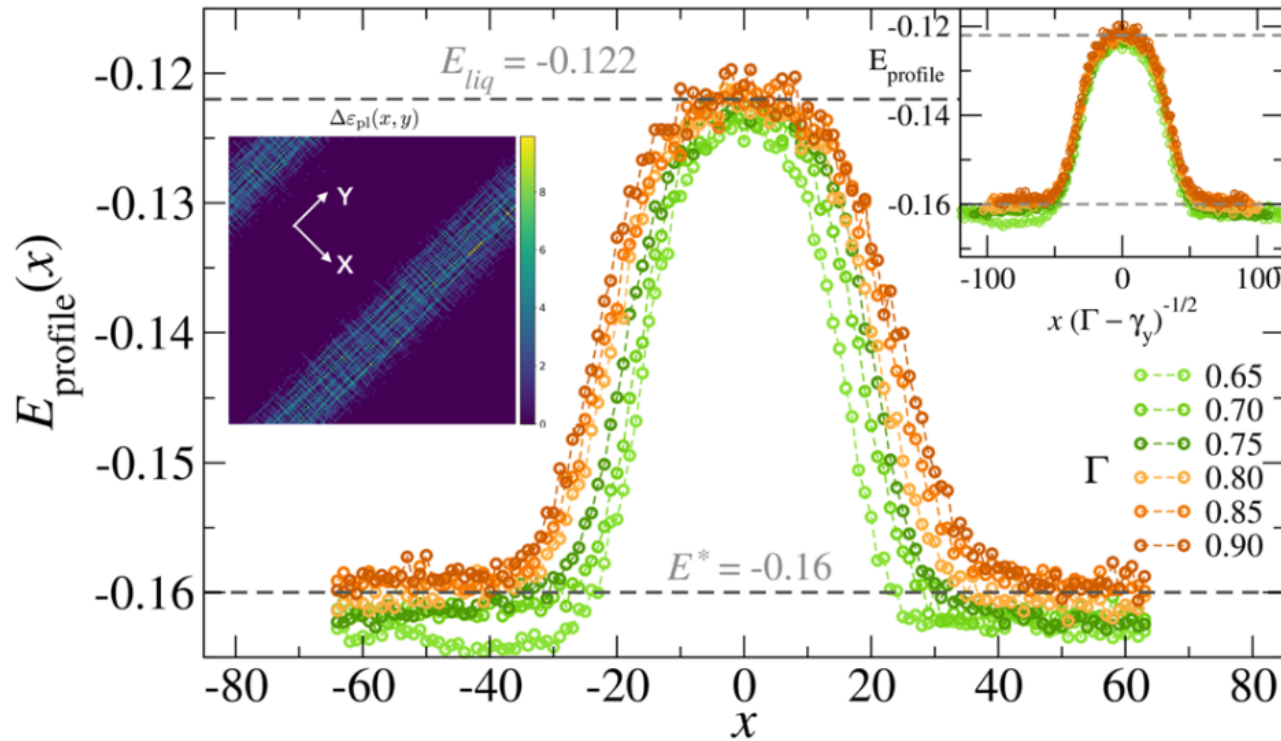
Finally,  **$E_{sf}$**  grows with  $\Gamma$  (orange line)



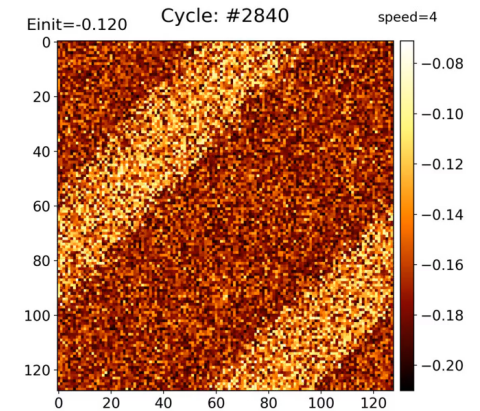
# Steady-state: "Phase diagram"



# Mixed state at $\Gamma > \Gamma_c$ : fluid band + marginal solid



## stroboscopic video



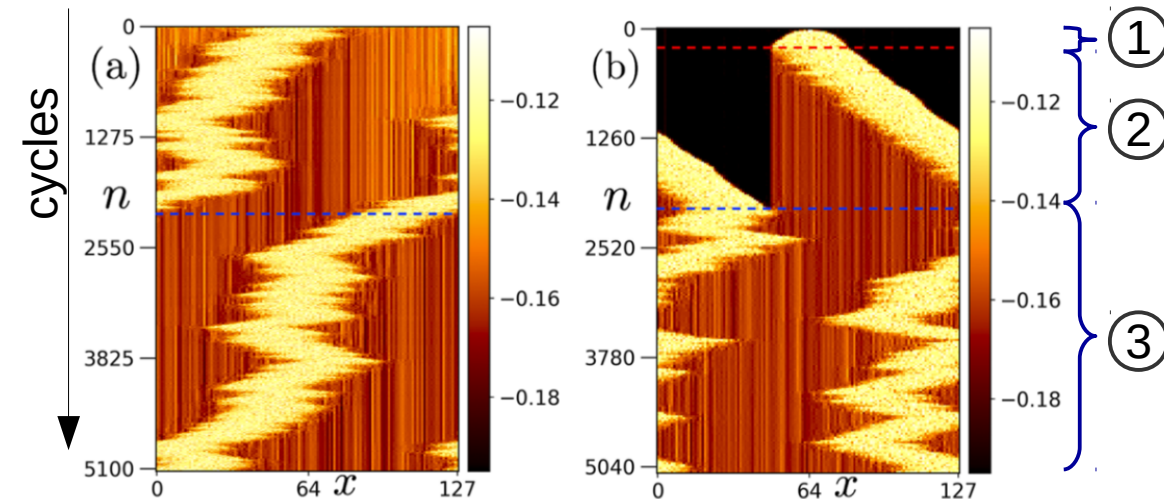
color: local energy

- **Center of the band** has the same energy as the **fluid** state in uniform deformation  $E_{\text{liq}}$
- **Outside** the band the system is mostly at the **marginal solid** state  $E^*$
- **Band-width increases** as  $w_s \sim (\Gamma - \Gamma_0)^{1/2}$ , with  $\Gamma_0 = \gamma_y$ .
- Notice that  $\gamma_y < \Gamma^* \leq \Gamma_c$ . So, **at the transition the band is finite**:  $w_s(\Gamma_c) > 0$

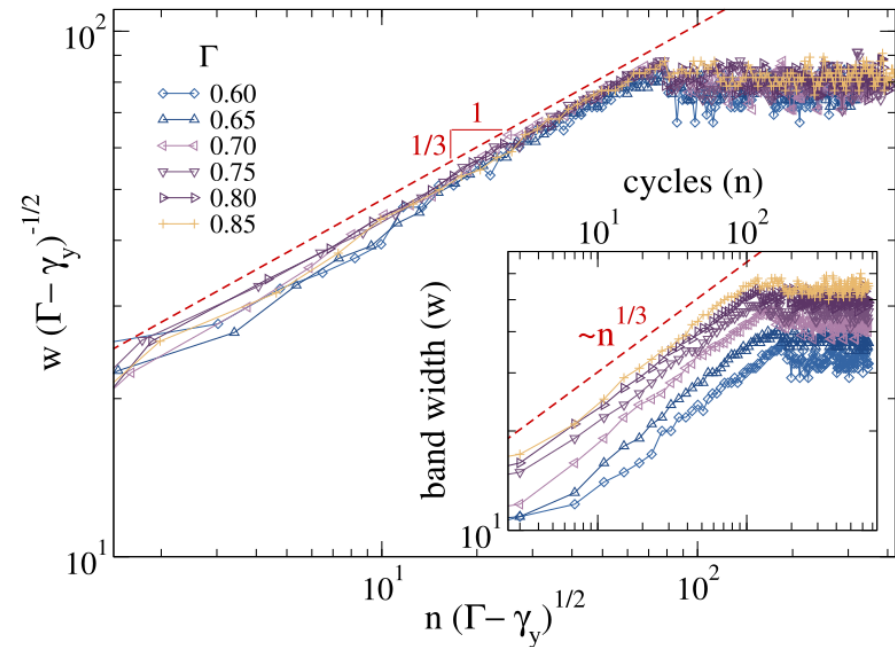
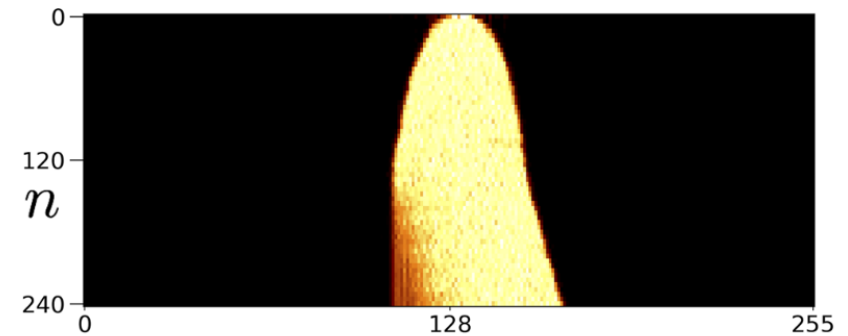
# Transient (at $\Gamma > \Gamma_c$ ): various dynamical stages

Poorly annealed

Well annealed



x: band-transverse coordinate; color: local energy

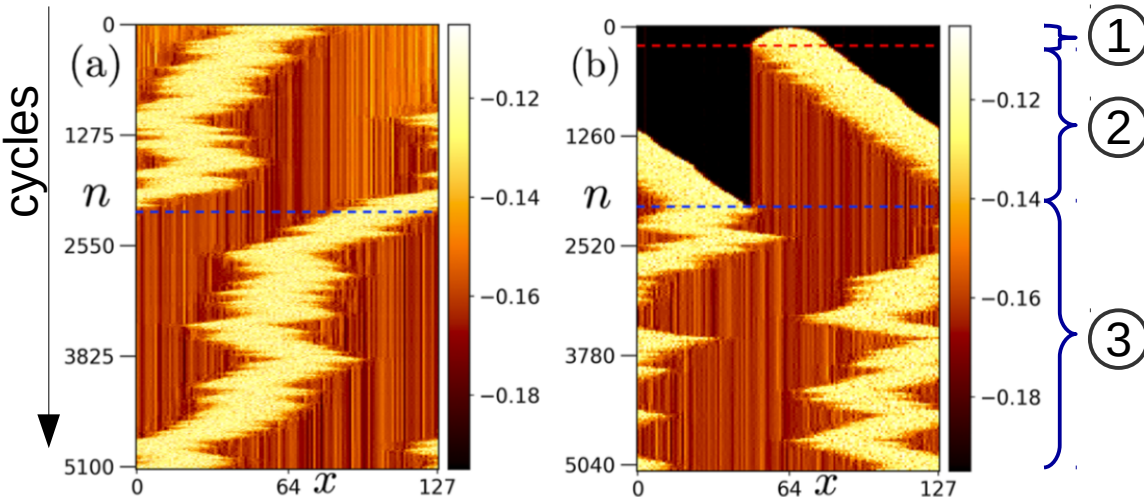


- 1) Band emergence and **coarsening**
- 2) Ballistic **swipe out** of deeply annealed region

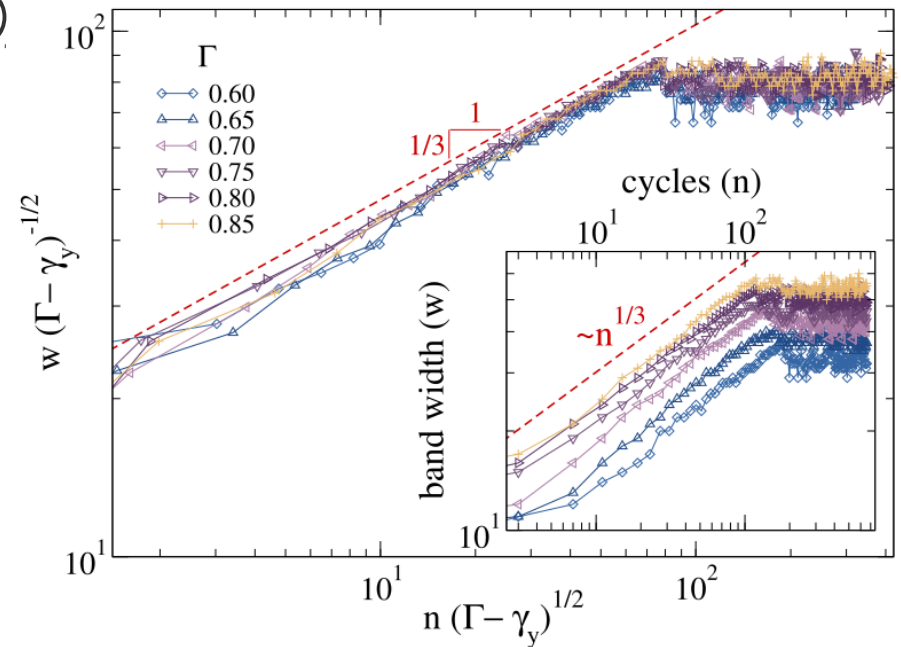
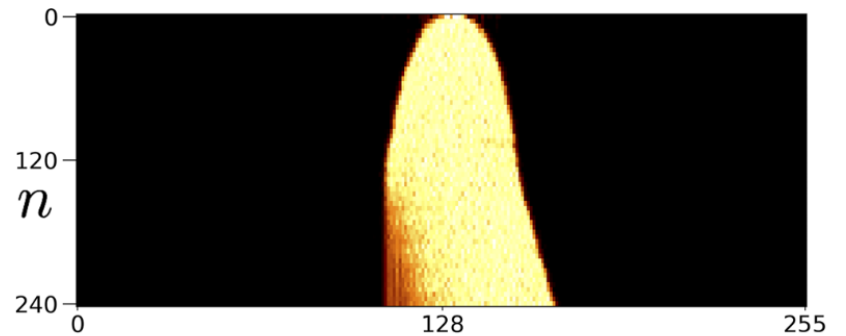
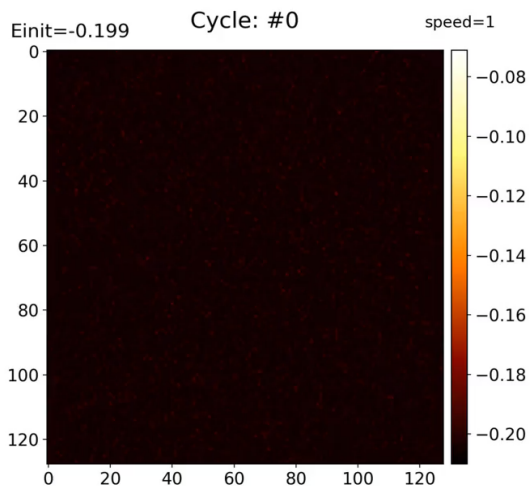
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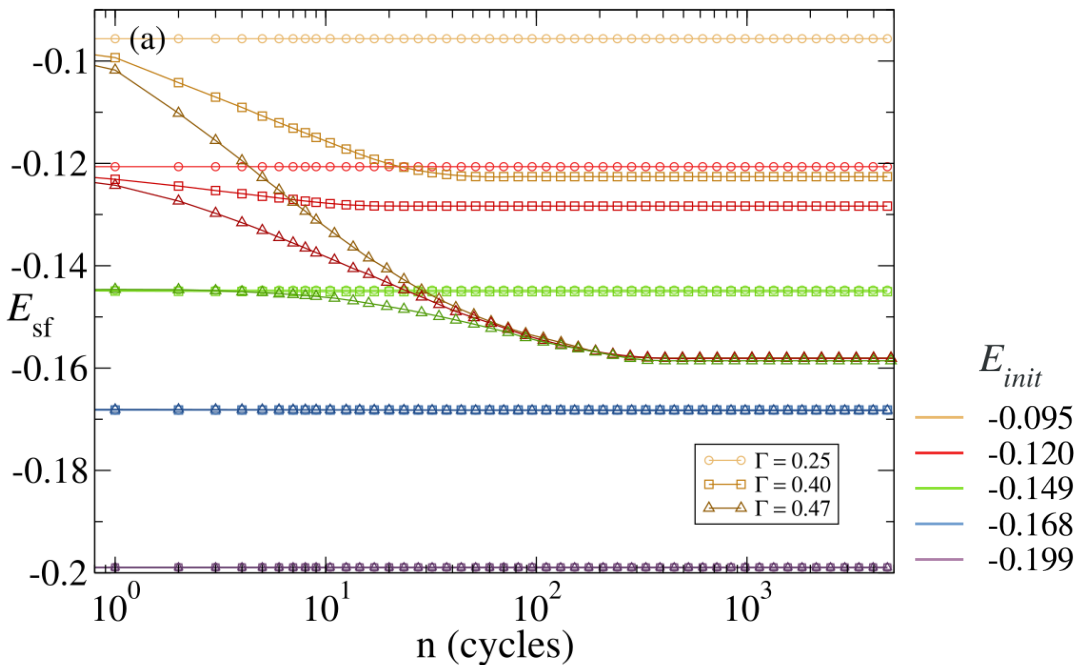
x: band-transverse coordinate; color: local energy



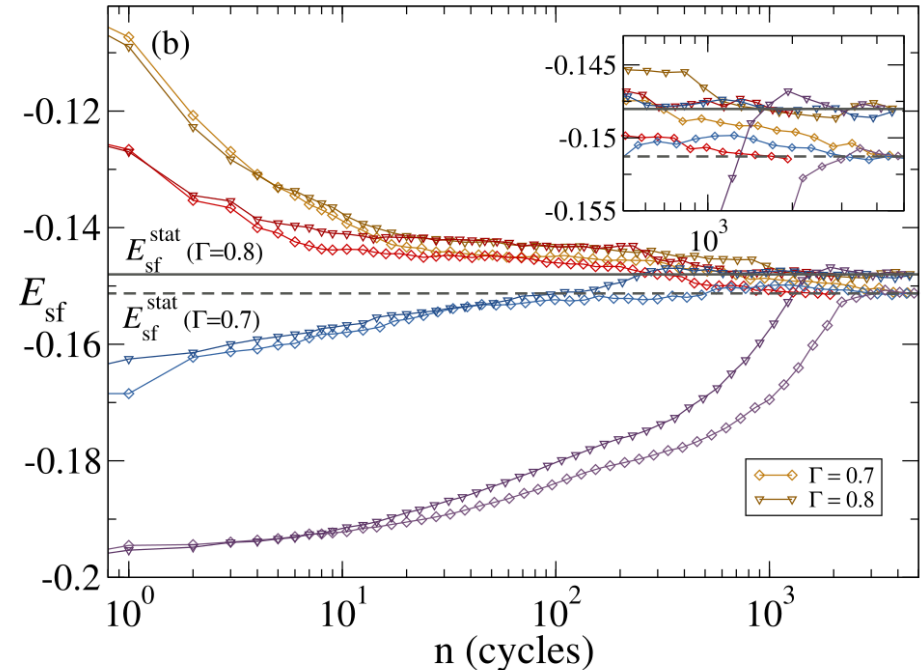
- 1) Band emergence and **coarsening**
- 2) Ballistic **swipe out** of deeply annealed region
- 3) Anomalous **diffusion**

# Transient behavior

$\Gamma < \Gamma_c$



$\Gamma > \Gamma_c$



**Ultra-stable systems** ( $E_{init} < E^*$ ) are unperturbed by the oscillations

Systems with  $E_{init} > E^*$  **shear-anneal** (transient annealing!) if the amplitude is large enough

**Notice** all three poorly annealed samples coming together to the same steady-state for  $\Gamma_a < \Gamma = 0.47 < \Gamma_c$

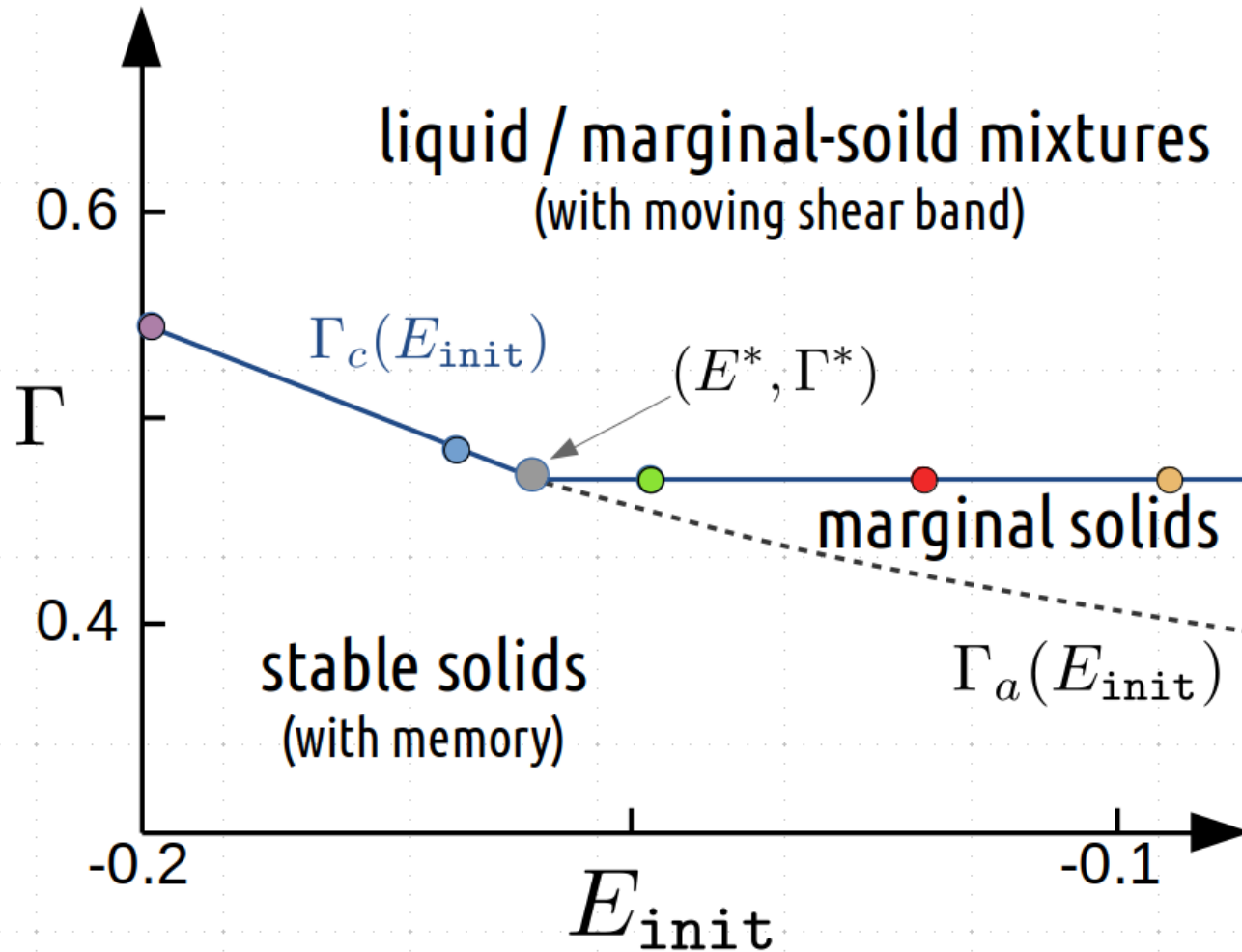
Above  $\Gamma_c$ , all samples go to the same ( $\Gamma$ -dependent) **stationary state at large  $n$**

The stationary  $E_{sf}$  grows with  $\Gamma$ , as the liquid **shear-band** of the mixed phase gets wider and wider



# Thanks!

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Have a look to ...



# Open questions/paths

- Are there transient states of **multiple** shear-bands in very large systems? How do they merge (or not) in a single band in the steady-state?
- If we set-up a **hierarchical disorder** with memory at small scales and no-memory at the meta-basin scale, do we recover the full atomistic simulations picture (including “reversible” plastic events)?
- How does the **critical annealing level  $E^*$**  detected in the oscillatory protocol compares to annealing temperatures that distinguishing between **ductile and brittle** yielding in uniform shear deformation?
- How do we explain the **ballistic motion** of the shear band observed for initially very deeply annealed systems?