



Some results on Plasticity in Amorphous Materials

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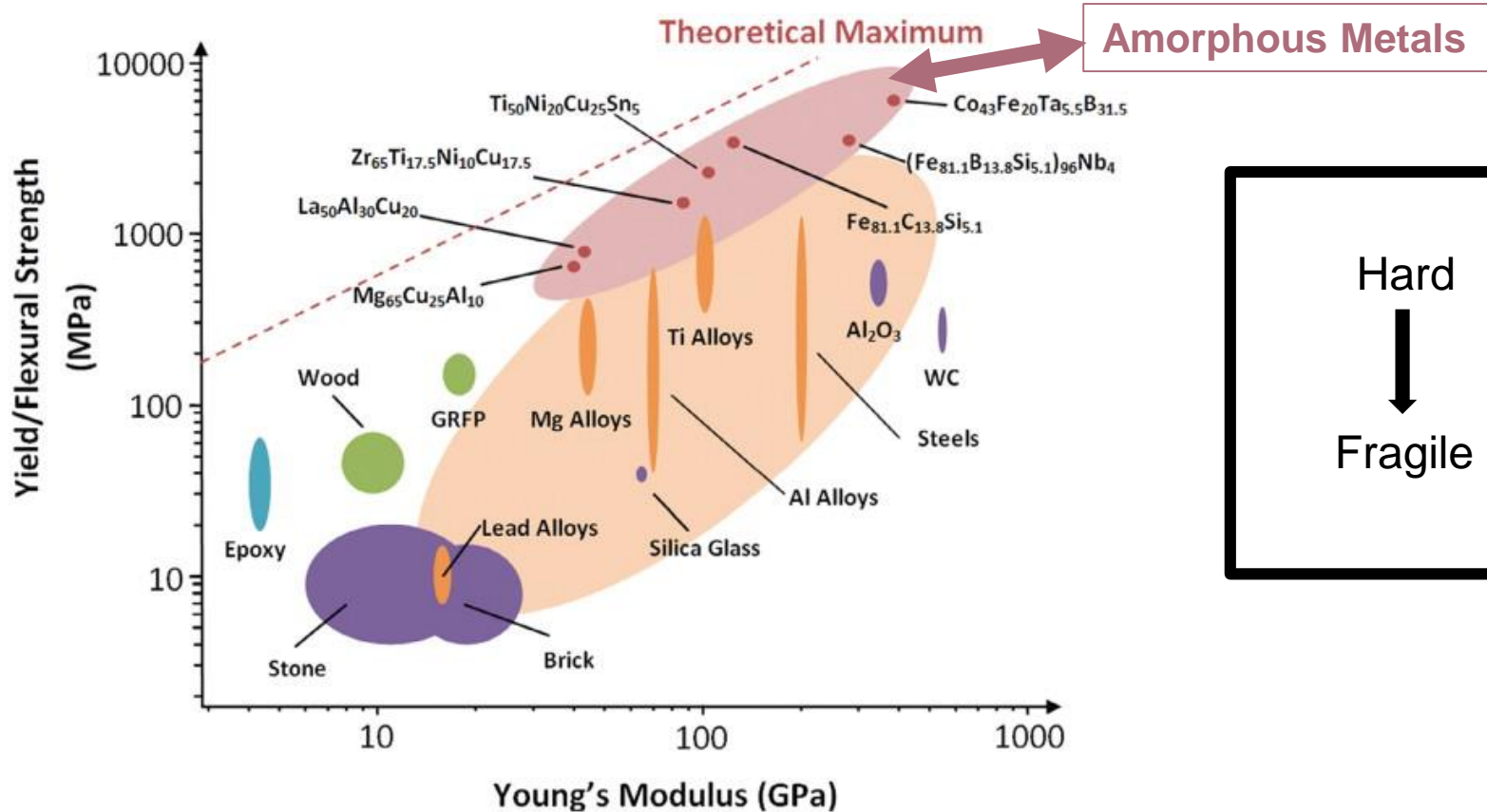
- I. Phenomenology**
- II. Mechanical Instability**
- III. Plastic Flow**



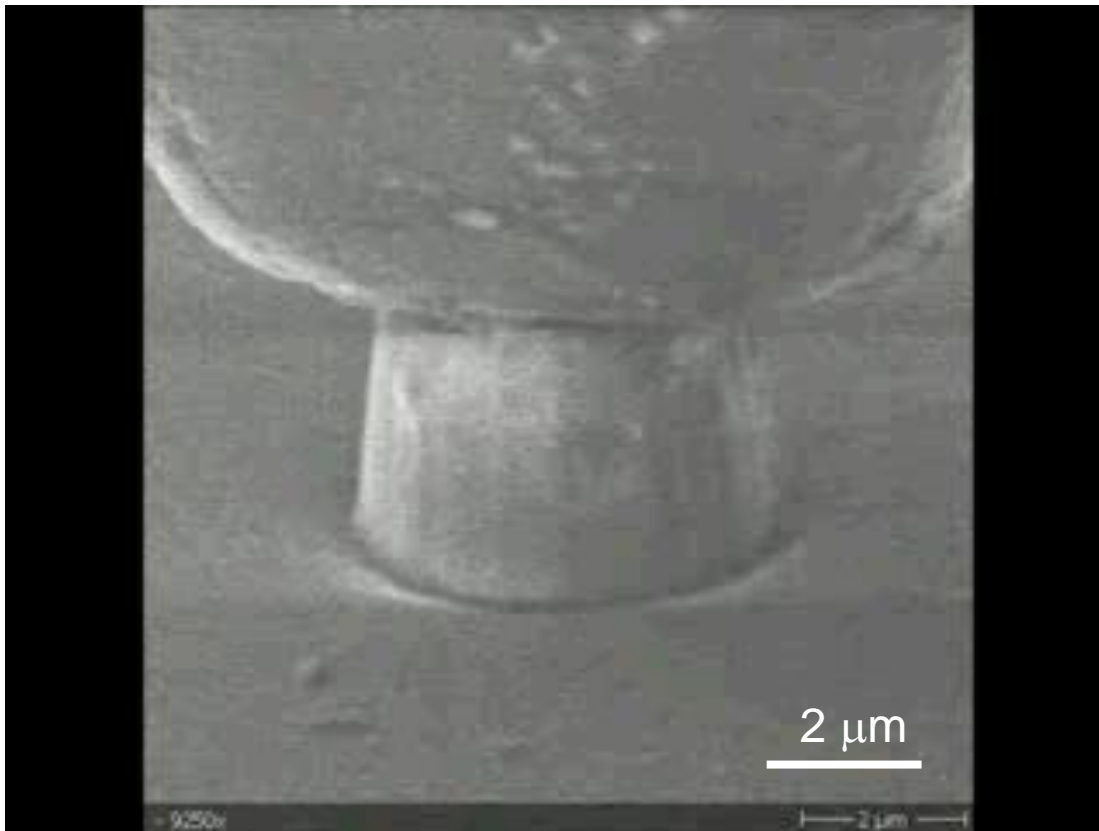
- I. Phenomenology**
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Amorphous materials are Very hard materials



but they are **Ductile** at **small scale**



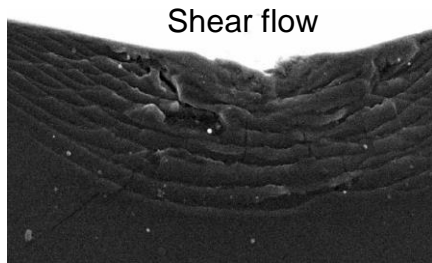
G. Kermouche, E. Barthel (2015)

Micropillar in pure silica glass

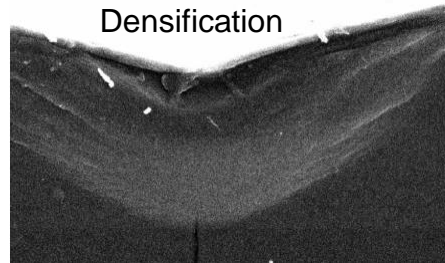


Their behaviour is **composition** and **loading** dependent

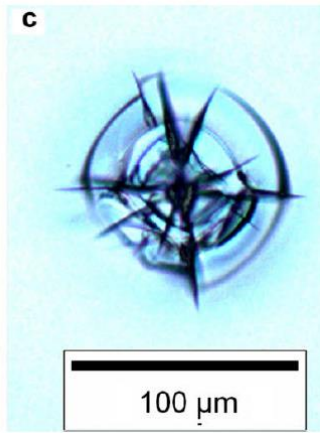
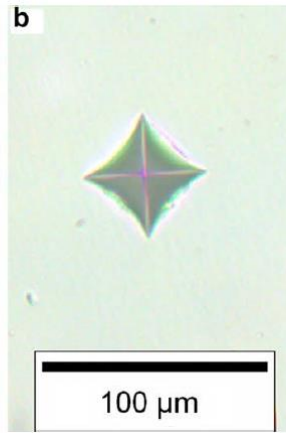
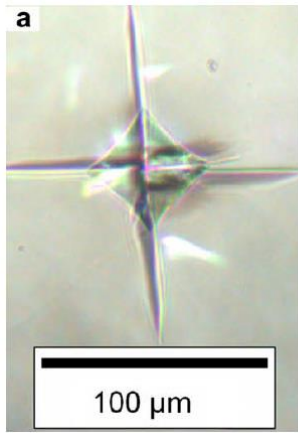
Composition



Soda-lime glass

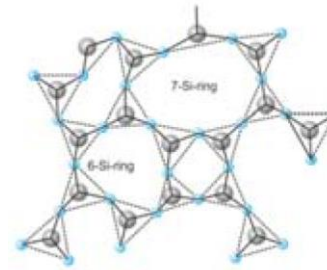


Anomalous glass

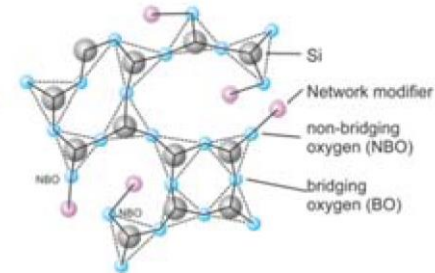


(a) 60% SiO₂ 20% Al₂O₃ 20% CaO; (b) 80% SiO₂ 10% Al₂O₃ 10% CaO; (c) 100% SiO₂

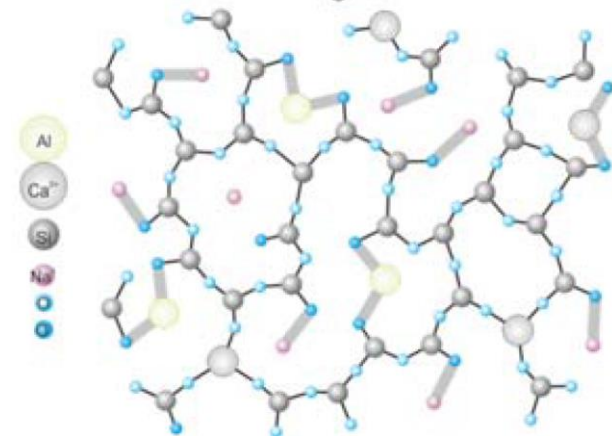
silica glass



sodium silica glass
(soluble glass)



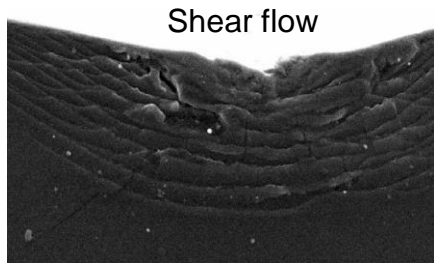
soda-lime silica glass



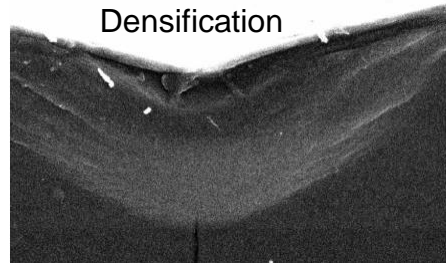
T.M. Gross et al. (2008, 2009)

Their behaviour is **composition** and **loading** dependent

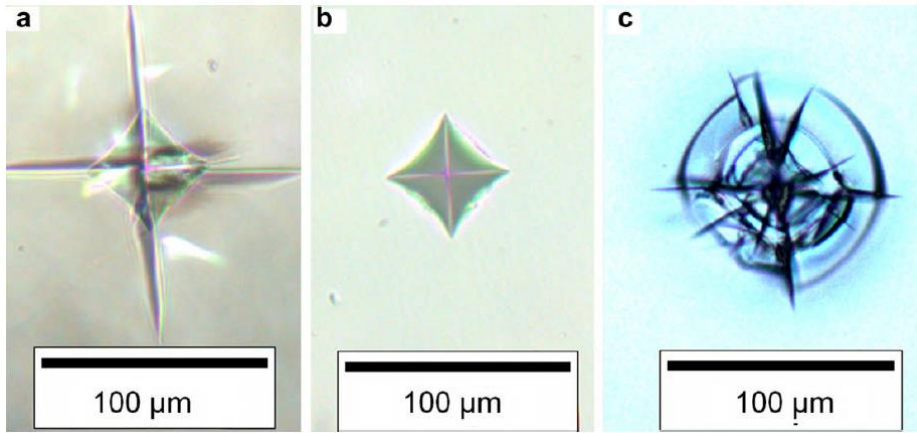
Composition



Soda-lime glass



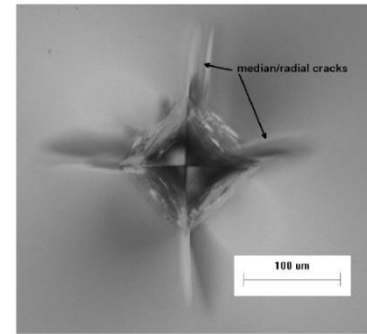
Anomalous glass



(a) 60% SiO₂ 20% Al₂O₃ 20% CaO; (b) 80% SiO₂ 10% Al₂O₃ 10% CaO; (c) 100% SiO₂

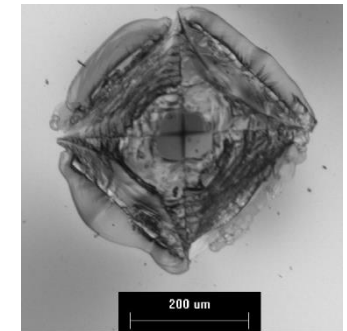
T.M. Gross et al. (2008, 2009)

Quasi-static
indentation
(0.2 mm/mn, 69N)



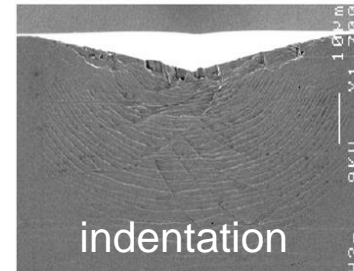
Strain Rate

Impact velocity
(410 mm/s, 562N)

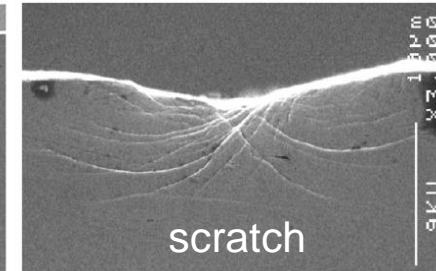


T.M. Gross et al. (2013)

Applied Load



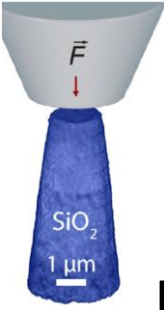
Bulk Metallic Glass



V. Keryvin et al. (2008)

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Their behaviour is **temperature** and **loading** dependent



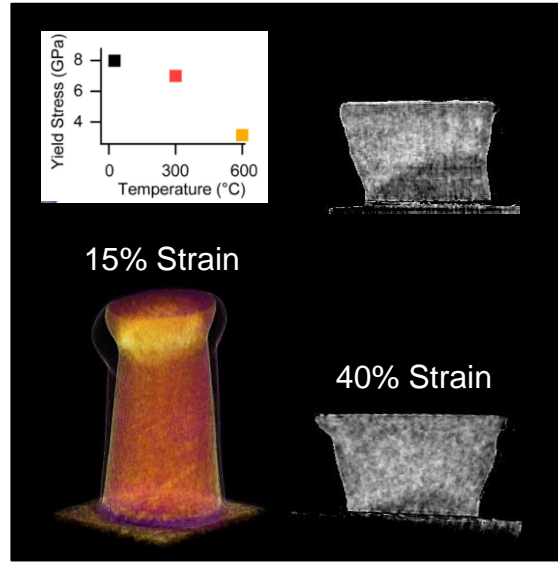
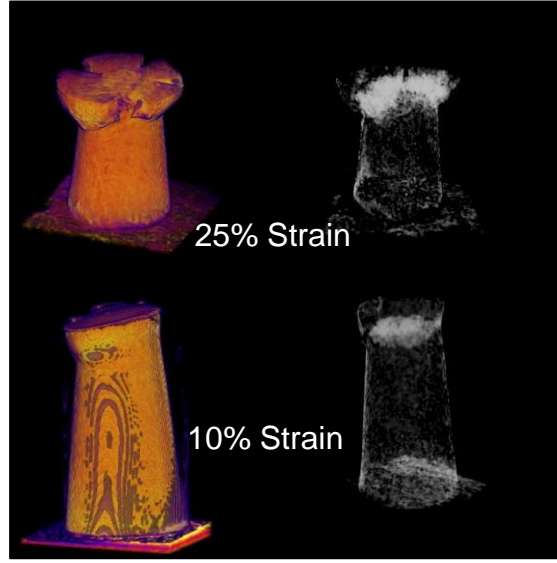
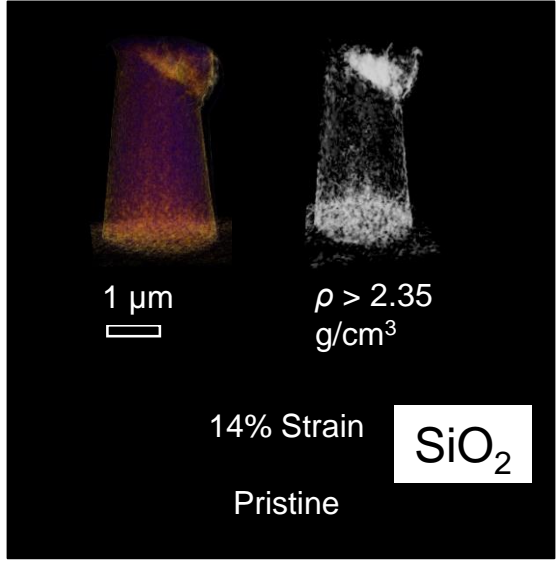
T=25°C

T=300°C

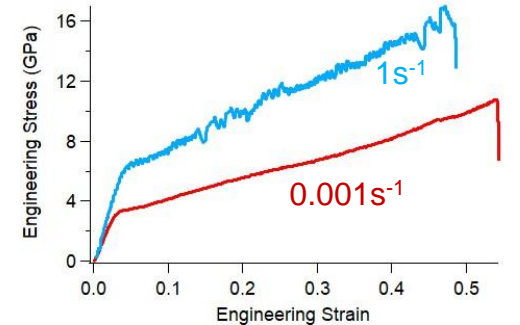
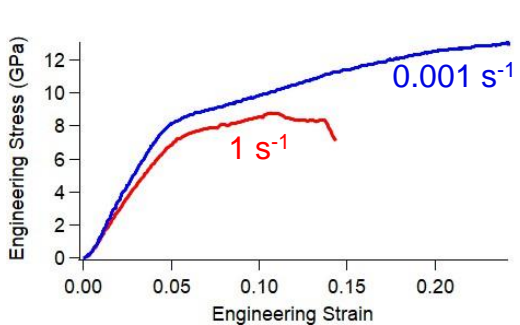
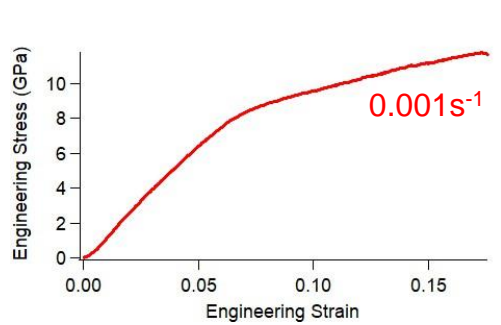
T=600°C

0.001 s⁻¹

1 s⁻¹



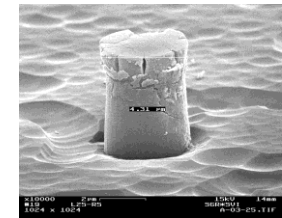
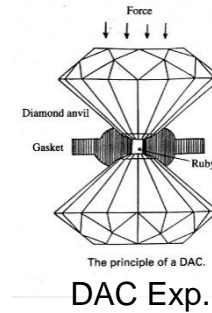
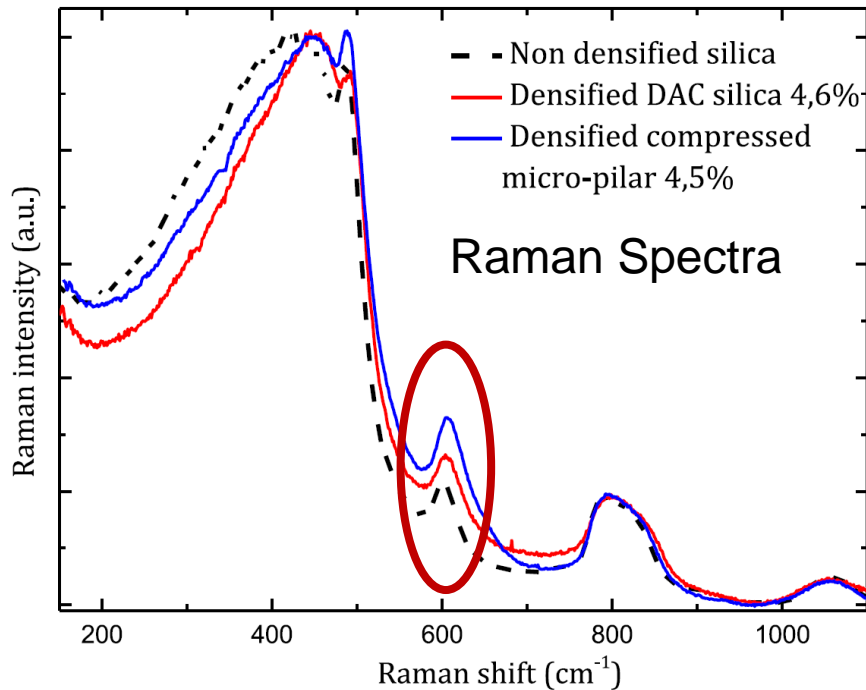
R.N. Widmer et al. (2022)



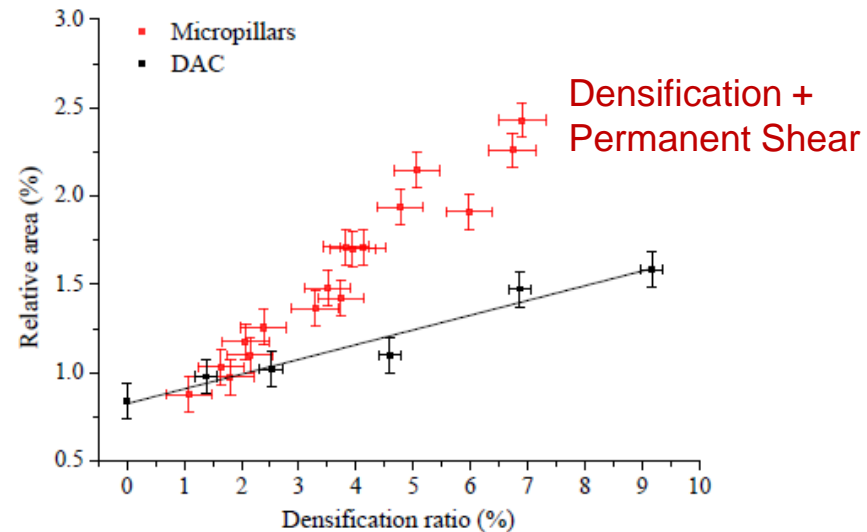


Their behaviour depends on the **loading history**

Comparison between different
densification processes in SiO_2



Micropillar Exp.



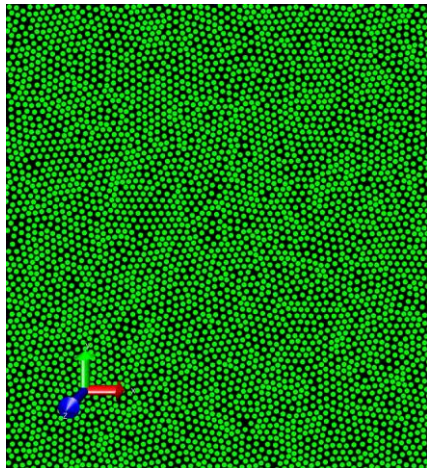


What happens **inside** a **disordered** material ?

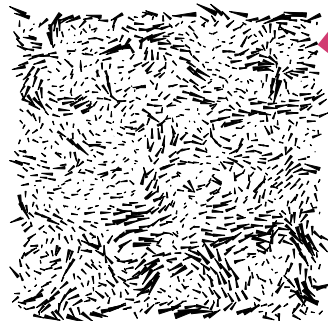


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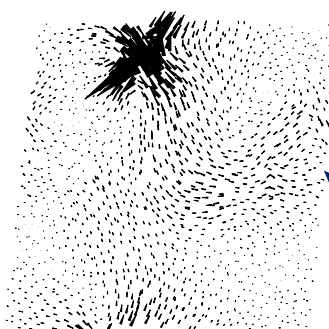
Non-affine displacement:



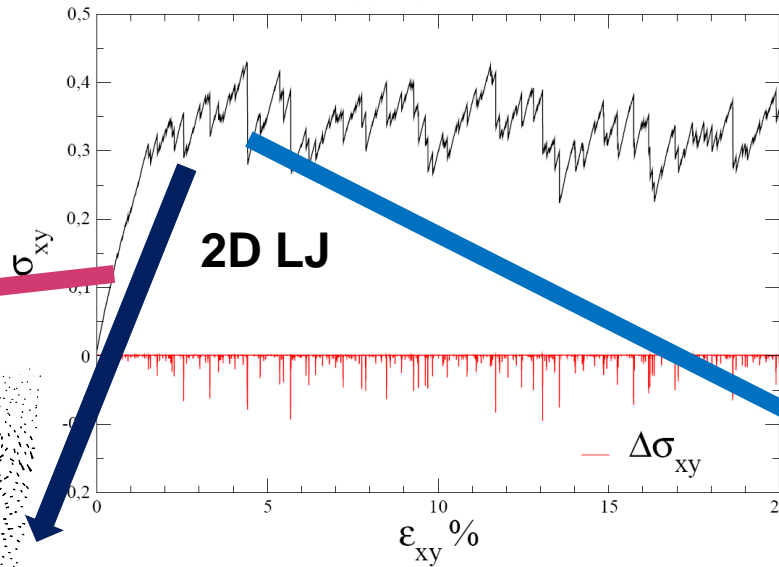
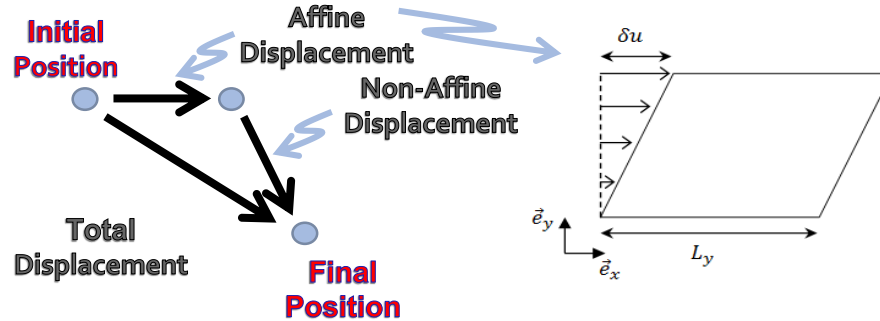
2D
LJ Glass



Non-affine reversible
displacements ($\times 10^3$)

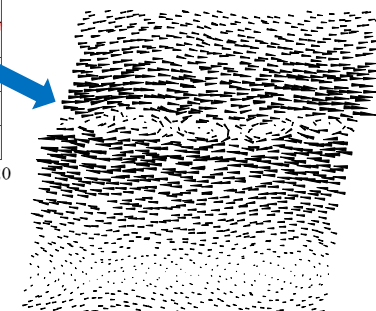


Local shear irreversible
($\times 40$) quadrupolar event



The strong localisation of the
deformation prevent stress losses

Athermal
Quasi-static
shear
deformation



Elementary
Shear band ($\times 0.4$)

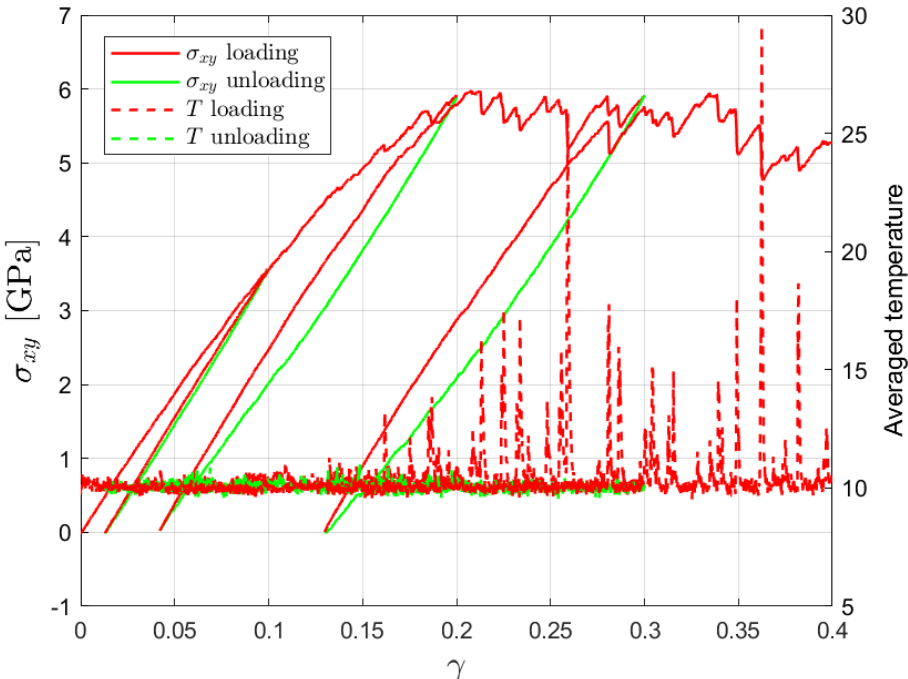


3D
CuZr Glass
 $T = 10\text{ K}$

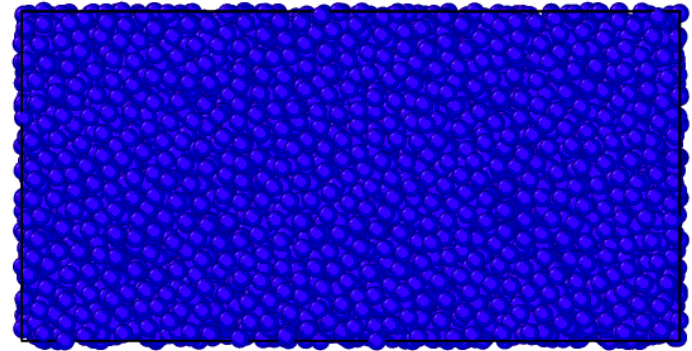
Local or collective **thermal** increases are a signature of a **collective instability**

Elementary vs. **permanent** shear bands

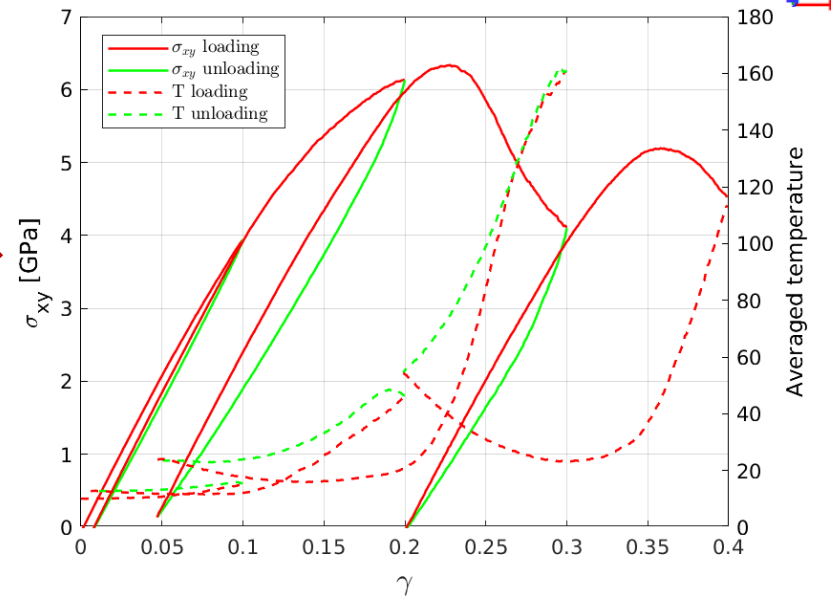
$$\dot{\gamma} = 1 \times 10^8 \text{ s}^{-1}$$



C1 LOADING



$$\dot{\gamma} = 1 \times 10^{10} \text{ s}^{-1}$$





- I. Phenomenology
- II. Mechanical Instability**
- III. Plastic Flow



At equilibrium, an **amorphous** solid is prepared in a **metastable** state.
A **mechanical equilibrium** corresponds to a local minimum of the **potential energy**.
At fixed **temperature**, it is kept in a local minimum of the **thermodynamic potential**.
The external driving deforms the energy landscape, up to the loss of equilibrium

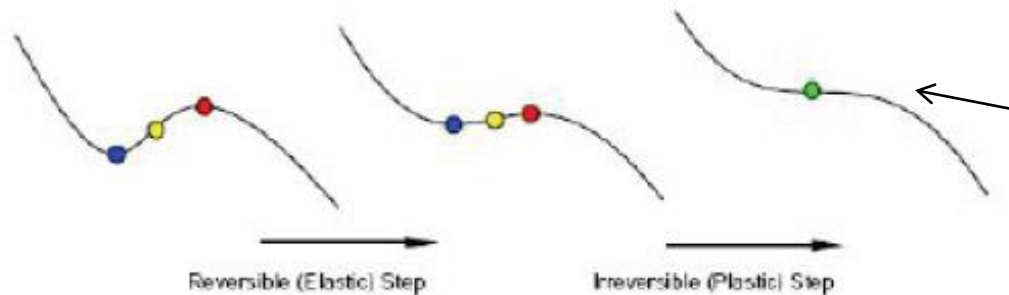
Hill's Criterion for Crystals Stability (1962) in Continuous Media

Helmholtz Free Energy $F(Y) \equiv F(X) + \Omega(X) \left\{ \tau_{ij}(X) \eta_{ij} + \frac{1}{2} C_{ijkl}(X) \eta_{ij} \eta_{kl} + \dots \right\}$

Stability Criterion

$$\min_{\{w,k\}} (C_{ijkl} w_i w_k k_j k_l) \geq 0$$

$\eta_{ij} \propto w_i k_j$
 ↑ ↑
 Burgers Vector Slip Plane





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Atomistic description in an **Amorphous material**:

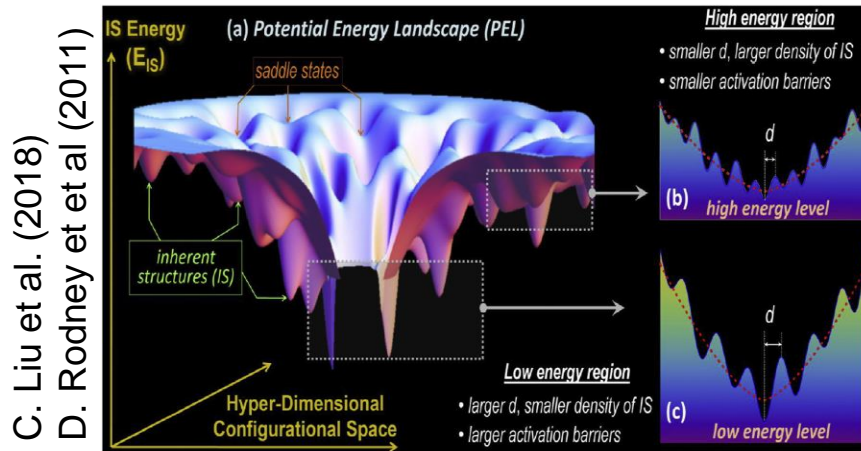
$$m_i \frac{\partial^2 u_\alpha}{\partial t^2}(\underline{r}_i, t) = - \sum_{j, \beta} M_{ij}^{\alpha\beta} \cdot u_\beta(\underline{r}_j, t) \quad \text{at } T=0K$$

with $M_{ij}^{\alpha\beta} = - \frac{\partial^2 E_{total}}{\partial r_{i\alpha} \partial r_{j\beta}}$ including external forces

$$\underline{V}(\underline{r}_i) e^{i\omega t} = \sqrt{m_i} \underline{u}(\underline{r}_i, t), \quad D_{ij}^{\alpha\beta} = \frac{M_{ij}^{\alpha\beta}}{\sqrt{m_i m_j}} \quad \text{Dynamical Matrix}$$

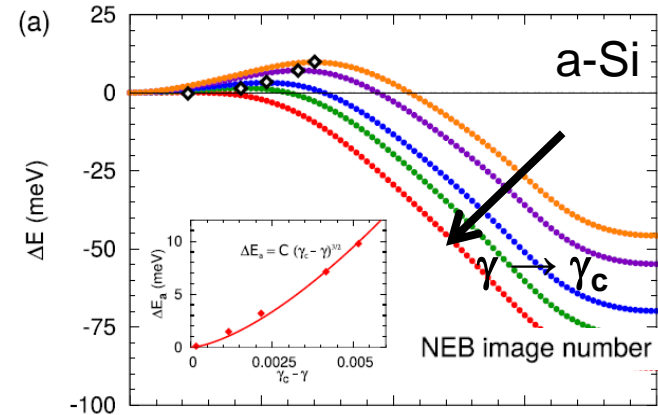
$$\omega^2 \underline{V} = \underline{\underline{D}} \cdot \underline{V} \quad \omega \rightarrow 0 \quad \text{Instability}$$

Potential Energy Landscape



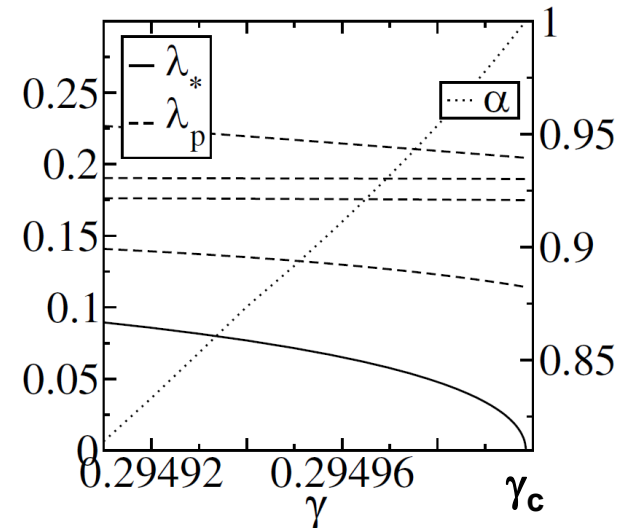
Energy Barriers

T. Albaret et al (2018)



Eigenvalues of the Dyn. Mat.

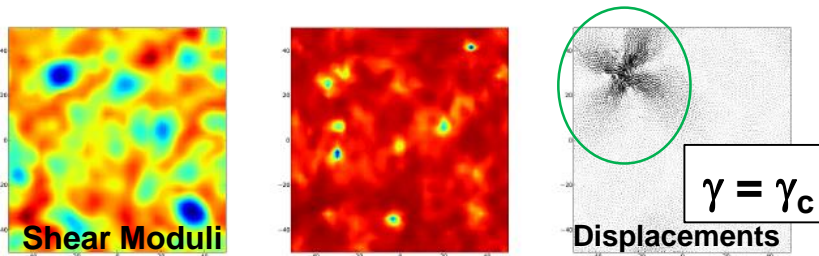
C. Maloney et al (2004)



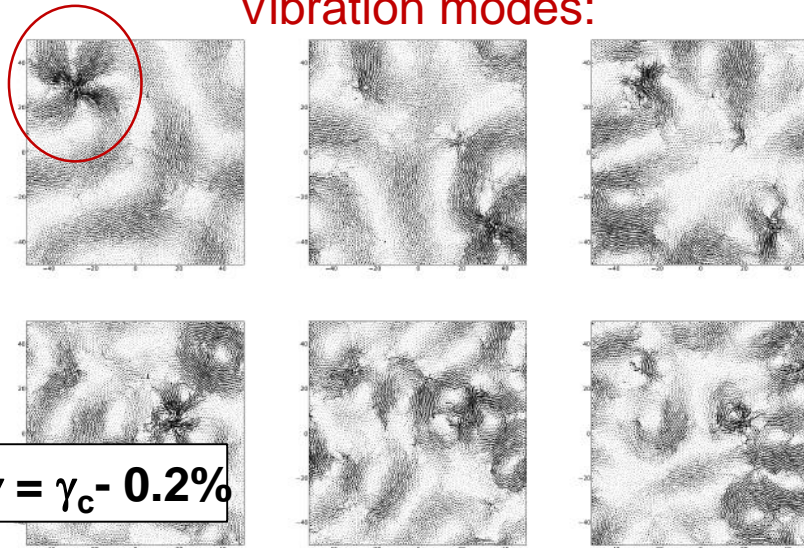


Prediction of plasticity through *soft* vibration modes and local elastic moduli in a 2D Lennard-Jones glass:

Quadrupolar Local Rearrangement (STZ):

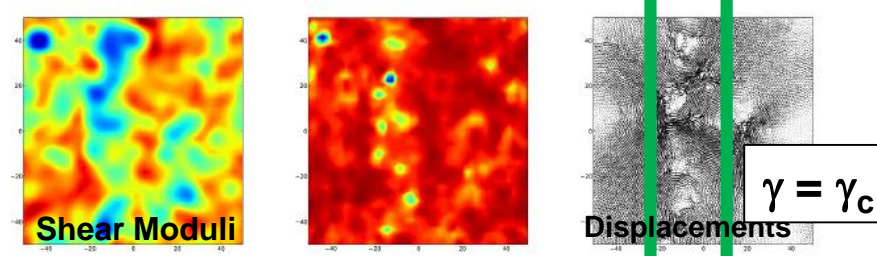


Vibration modes:

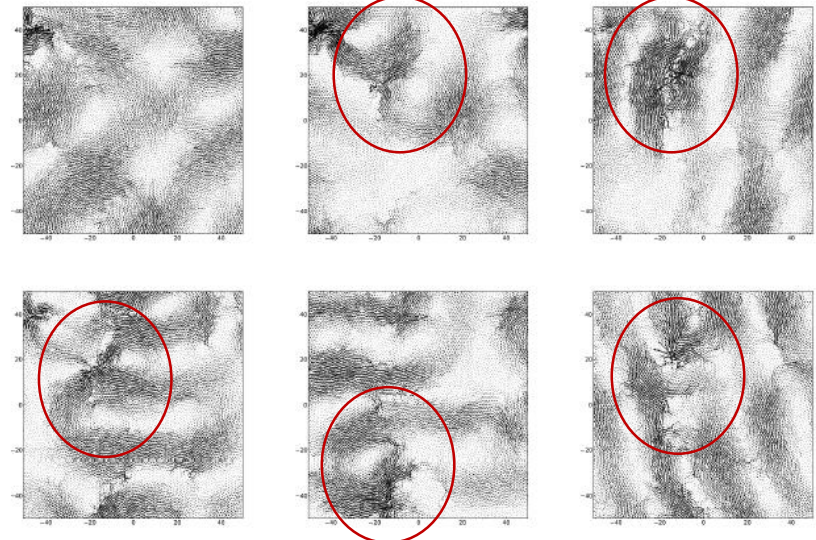


A single localized mode

Elementary Shear Band:



Vibration modes:



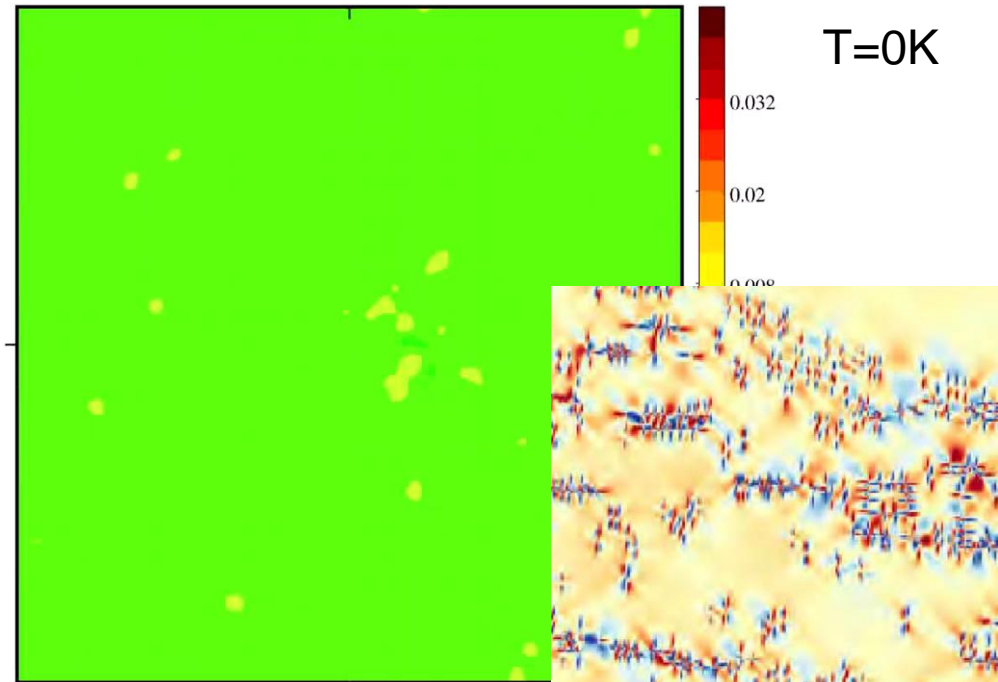
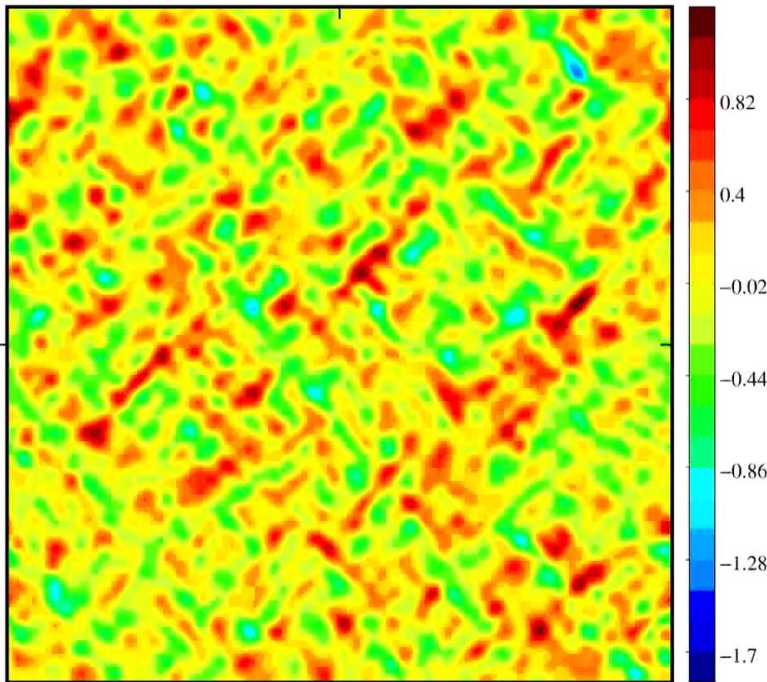
Superposition of localized modes, on percolating soft zones

A. Tanguy et al. (2010)

Signature of Plasticity in the local shear stress components in a 2D LJ Glass

Step= 2

Step= 2



Pristine micro-crystal
L. Truskinovsky et al. (2021)

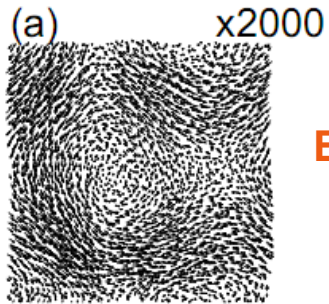
Total Local Shear Stress
Incremental + Quenched Stress

Incremental Local Shear Stress

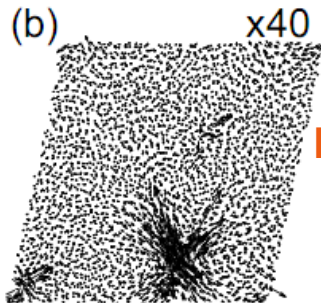
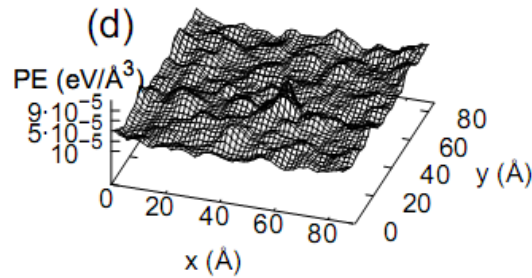
M. Tsamados et al. (2008)

Elementary Rearrangements in Plastic Deformation

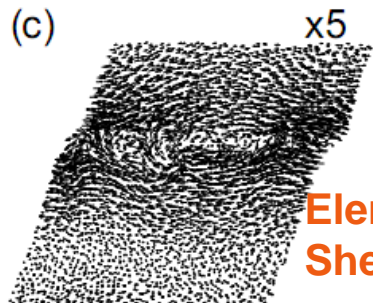
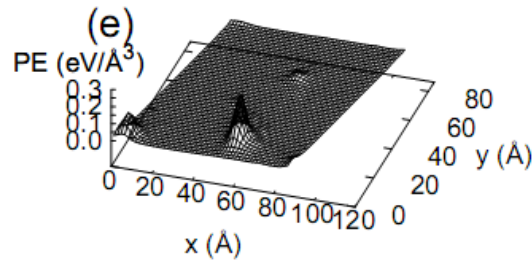
Athermal Quasi-Static deformation of a-Si



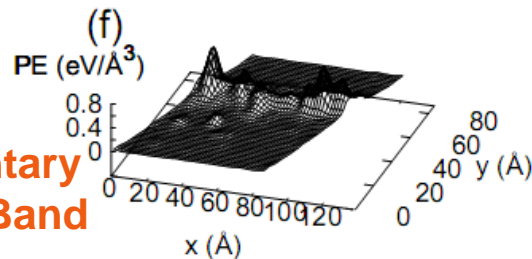
Elastic



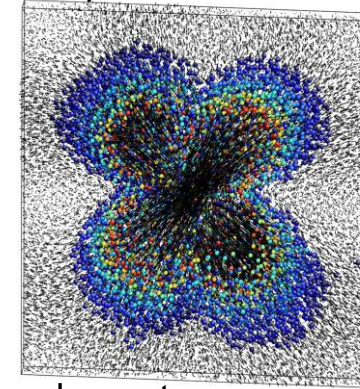
Plastic



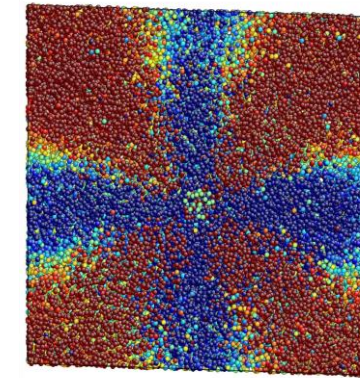
Elementary
Shear Band



displacements

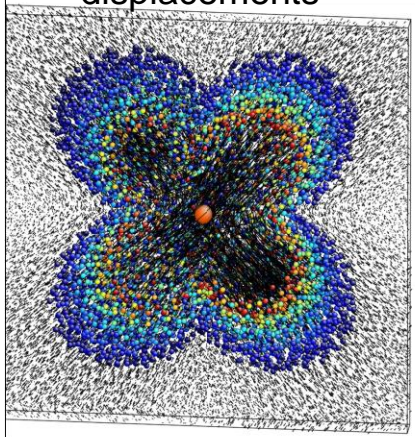


shear stress

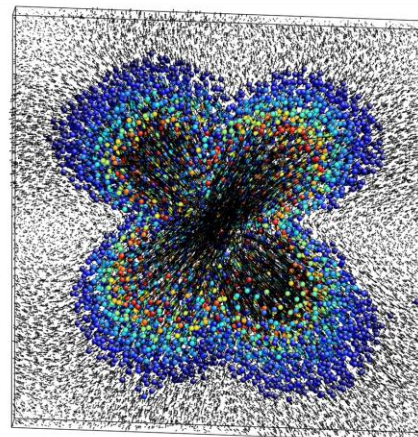
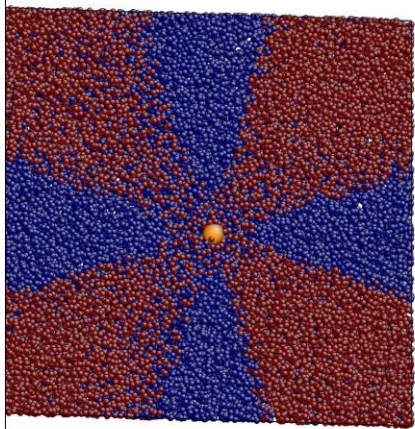


Local Irreversible Plastic deformation

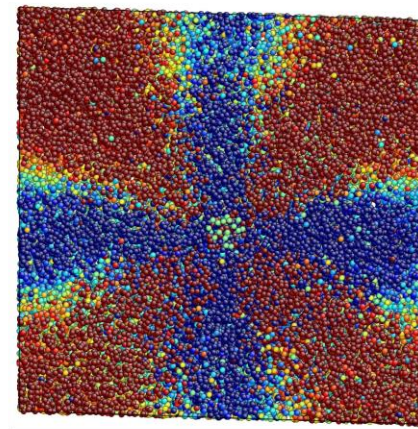
displacements



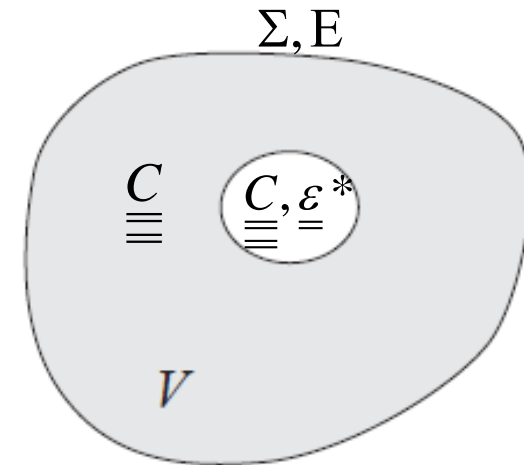
shear stress



measured



T. Albaret et al. (2016)



Eshelby-like Inclusion

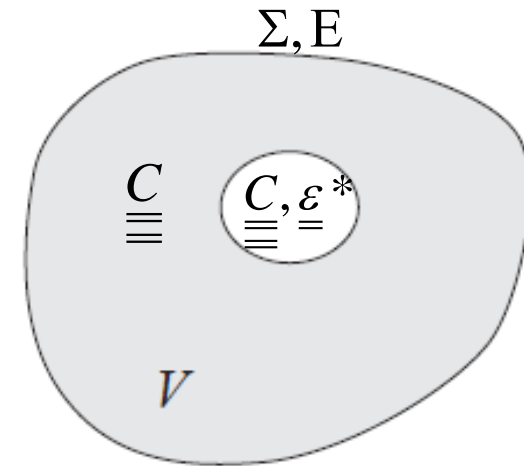
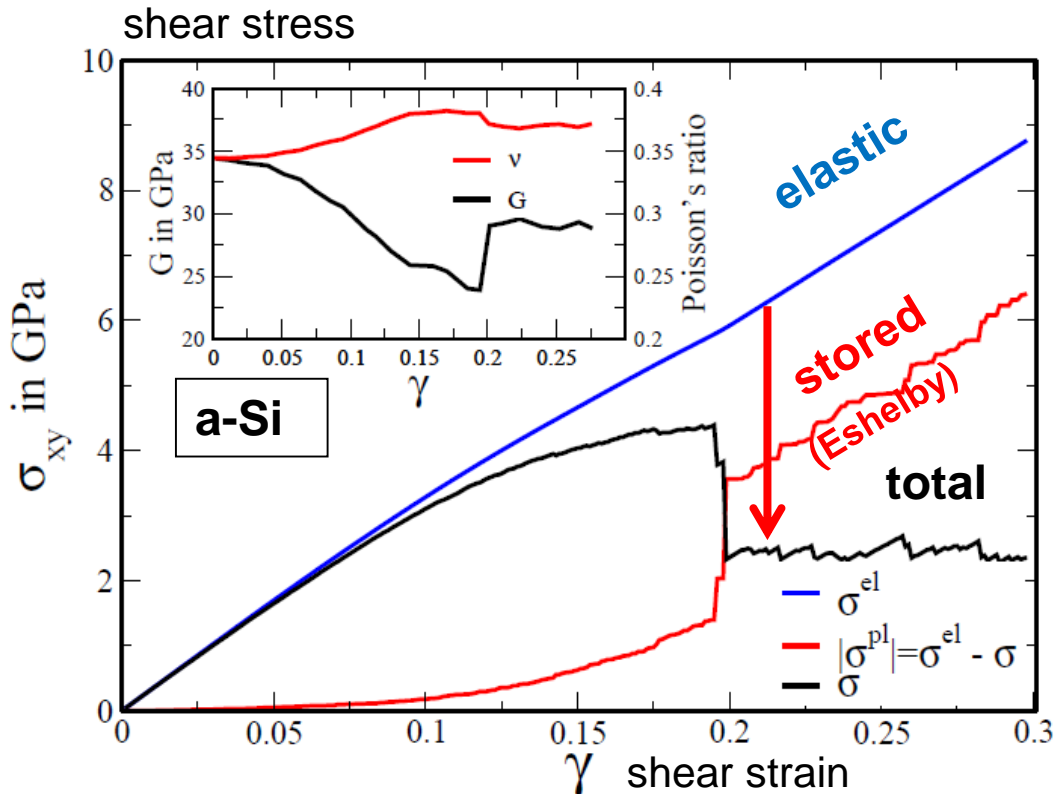
$$\operatorname{div} \underline{\underline{\sigma}}^I = 0 \Leftrightarrow \operatorname{div} \left(\underline{\underline{C}} : \underline{\underline{\varepsilon}} \right) + f(\underline{\underline{\varepsilon}}^*) = 0$$

$$f(\underline{\underline{\varepsilon}}^*) = \underline{\underline{C}} : \underline{\underline{\varepsilon}}^* \cdot \underline{\underline{n}} \delta(S^I) \Rightarrow \underline{\underline{\varepsilon}} = \underline{\underline{S}} : \underline{\underline{\varepsilon}}^*$$

T. Albaret et al. (2016)



Local Irreversible Plastic deformation

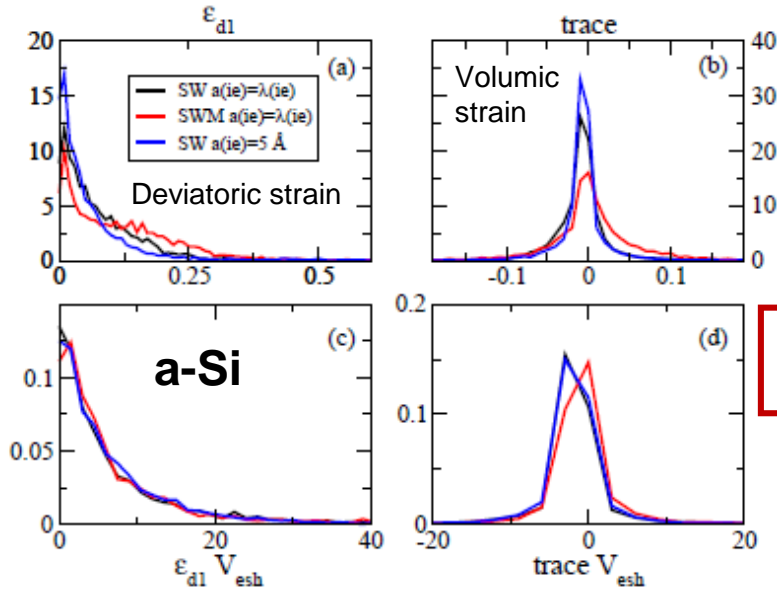


Eshelby-like Inclusion

$$\delta\sigma_{xy} = 2G(\gamma)\delta\varepsilon_{xy} - 2G(\gamma)\delta\varepsilon_{xy}^*$$



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Fitting Eshelby tensor vs. Constitutive Laws:
the example of a-Si

$$\delta_i \epsilon_{xy} = \delta_i \epsilon_{xy}^{el} + \delta_i \epsilon_{xy}^{pl}$$

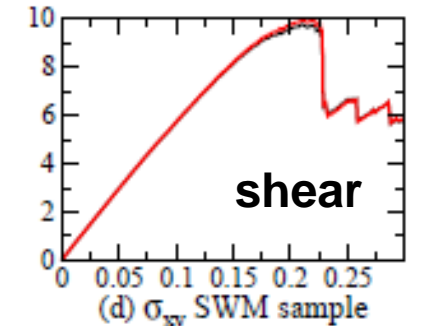
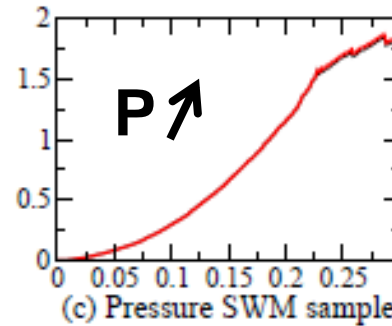
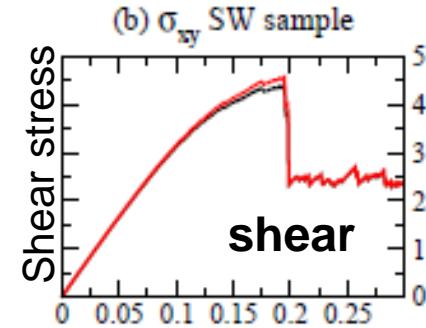
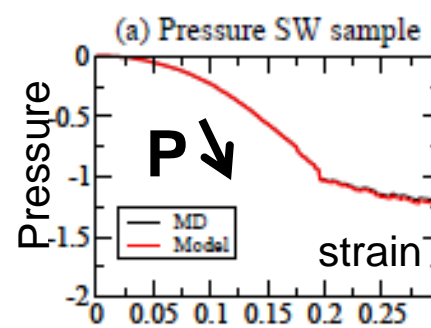
$$\delta_i \sigma_{xy} = 2G(\gamma_i) \delta_i \epsilon_{xy} - 2G(\gamma_i) \delta_i \epsilon_{xy}^{pl} = \delta_i \sigma_{xy}^{el} - \delta_i \sigma_{xy}^{pl}$$



Constitutive Law

Elementary bricks for Mesoscopic Modeling

T. Albaret et al. (2016)



Bonds Bending Rigidity ↓

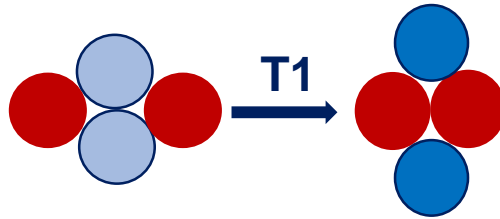


Example of a **2D Lennard-Jones** glass

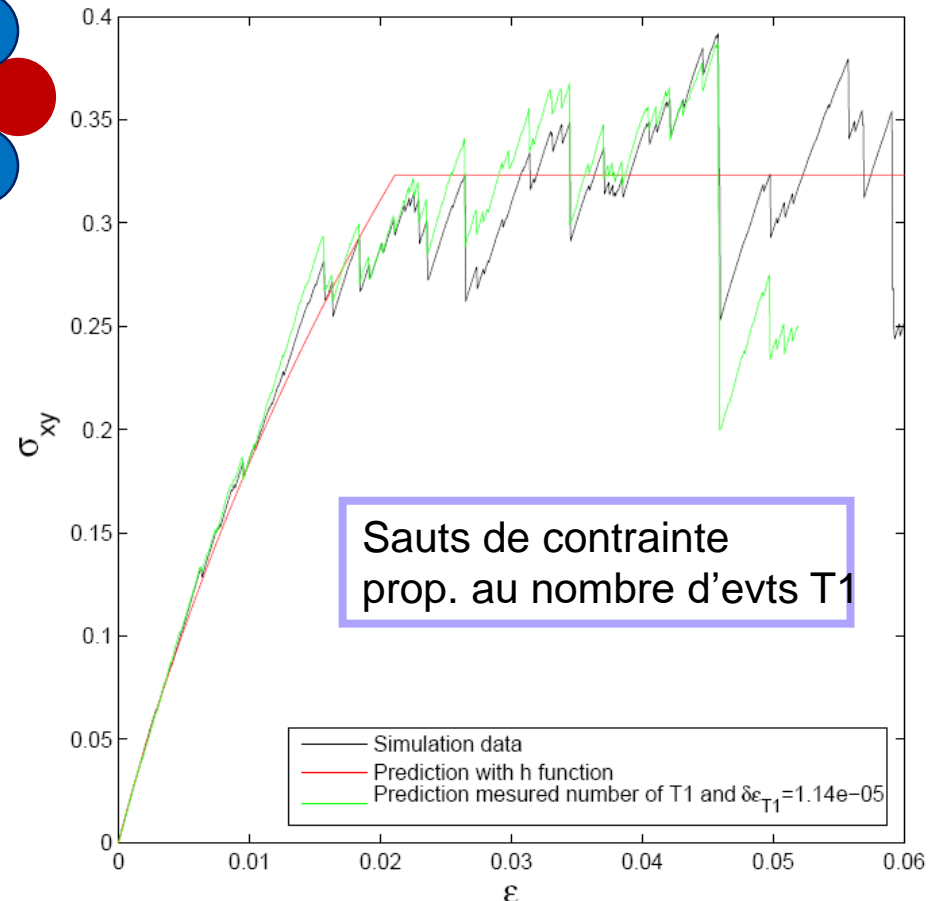
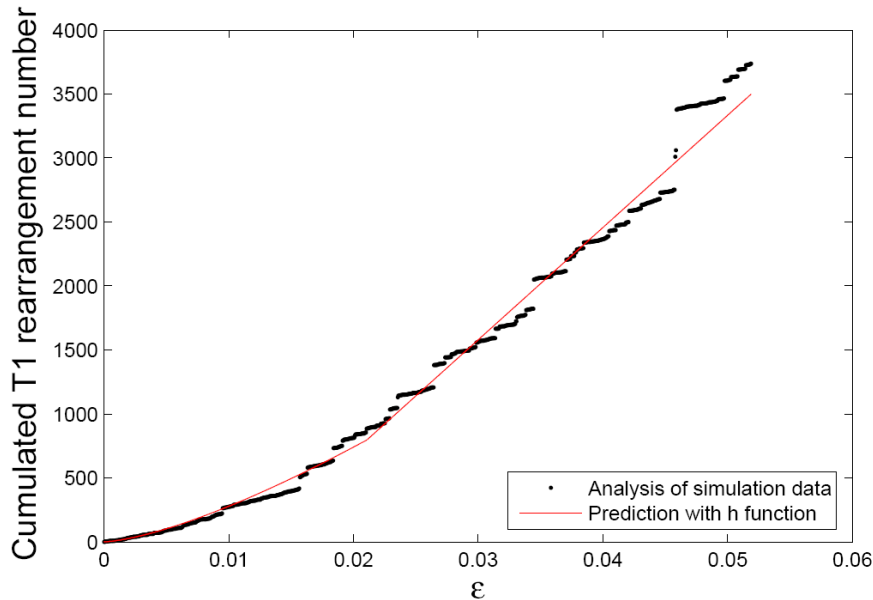
él. pl.

$$\dot{\epsilon} = \dot{U} + \dot{\epsilon}_p$$

$$\epsilon_p = N^{T1} \delta\epsilon^{T1} :$$



$$\sigma = \mu(\epsilon - N^{T1} \delta\epsilon^{T1})$$



T1 rearrangement in a 2D **silica glass**

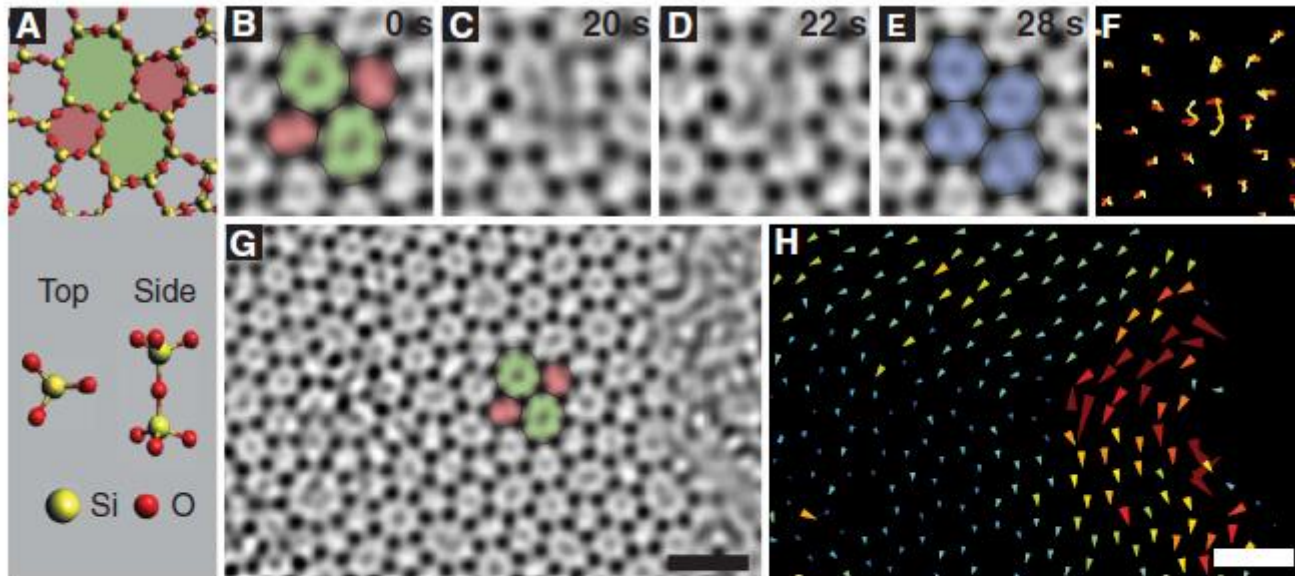
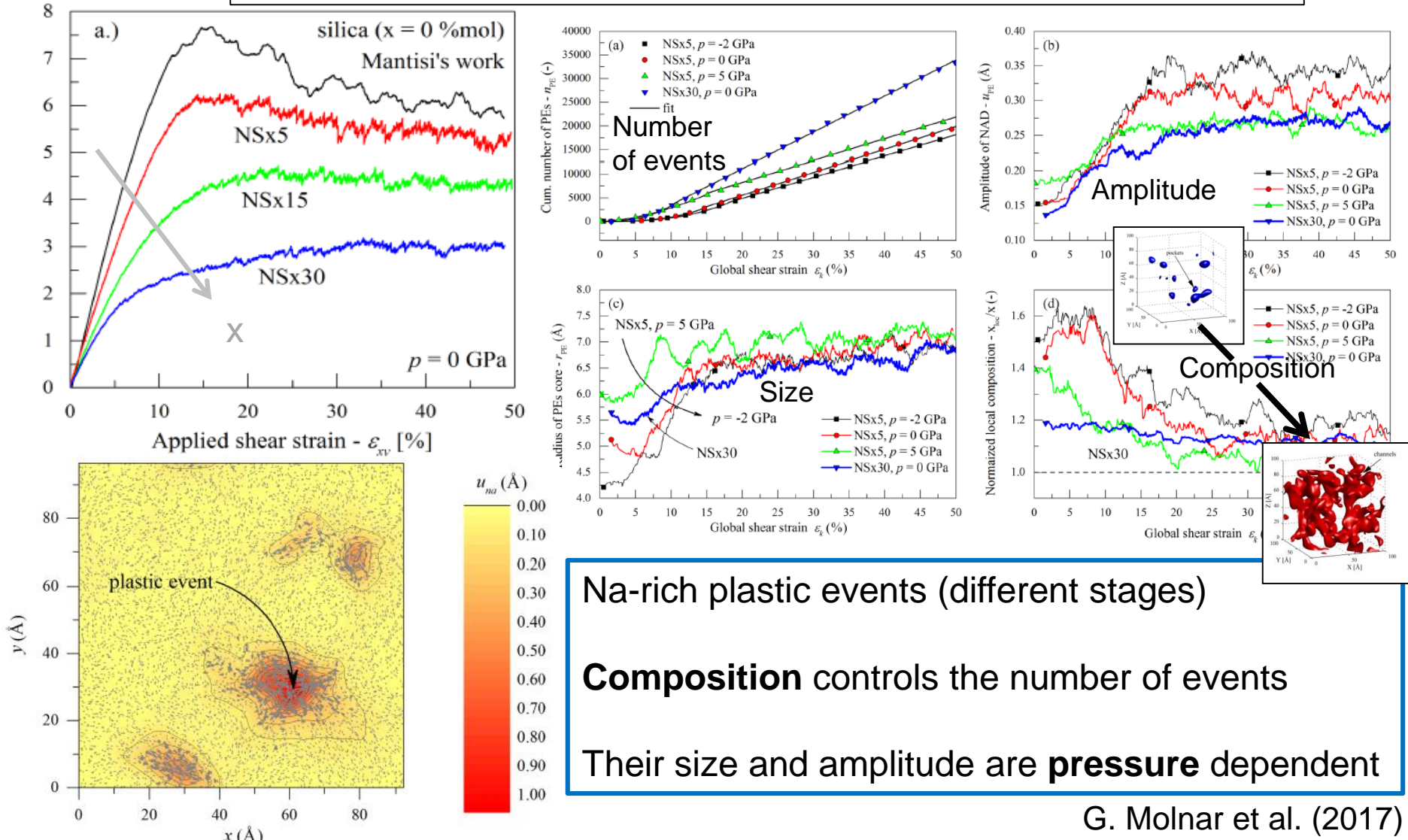


Fig. 1. Elastic and plastic deformation in ring exchange. (A) Cartoon models of the 2D silica structure. (B to E) TEM images showing a ring rearrangement that transforms a 5-7-5-7 cluster into a 6-6-6-6 cluster. The dark spots are Si-O-Si columns that correspond with the top and side views in (A). Images have been smoothed and Fourier-filtered to remove the graphene lattice background [see figs. S2 and S3 and (17)]. (F) A trajectory map of the atomic sites. Color (red to yellow) indicates time of motion. (G) Larger view of the region from (A), and (H) corresponding first-to-last frame displacement map. The arrows have been enlarged $\times 2$ to increase visibility; color indicates size of displacement, from 0 (dark blue) to ≥ 1.3 Å (red). The region between the bond rearrangement and the edge of the sheet exhibits strong local rotation. Scale bars: 1 nm. See also movies S1 and S2.

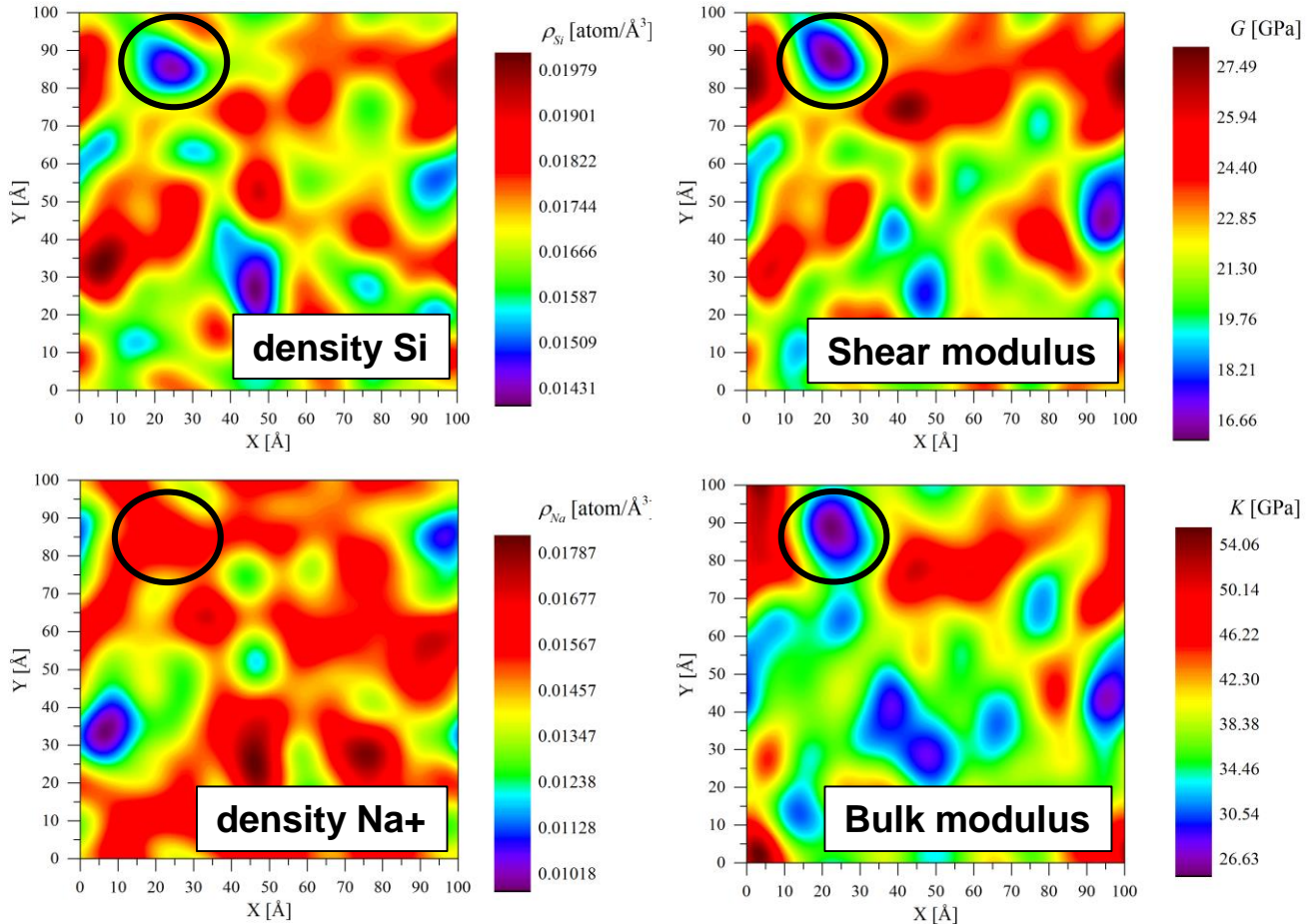
Sensitivity to **Composition** and **Pressure** in $(1-x) \text{SiO}_2 + x \text{Na}_2\text{O}$



Na-rich plastic events (different stages)
Composition controls the number of events
Their size and amplitude are **pressure** dependent



Elastic Moduli in a sodo-silicate glass $(1-x)\text{SiO}_2 + x\text{Na}_2\text{O}$:



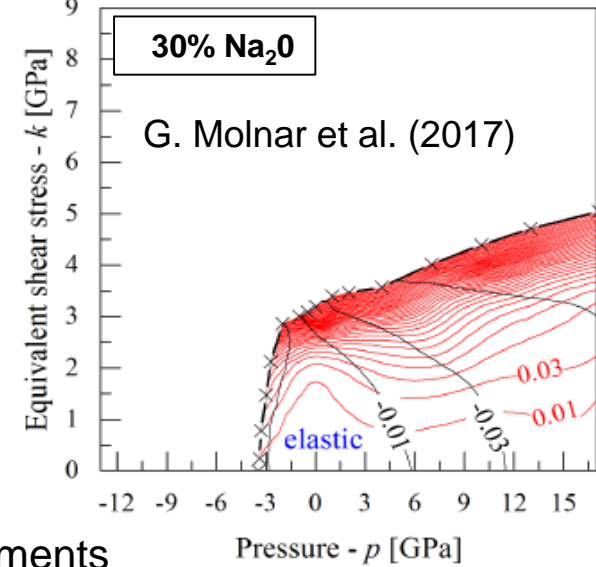
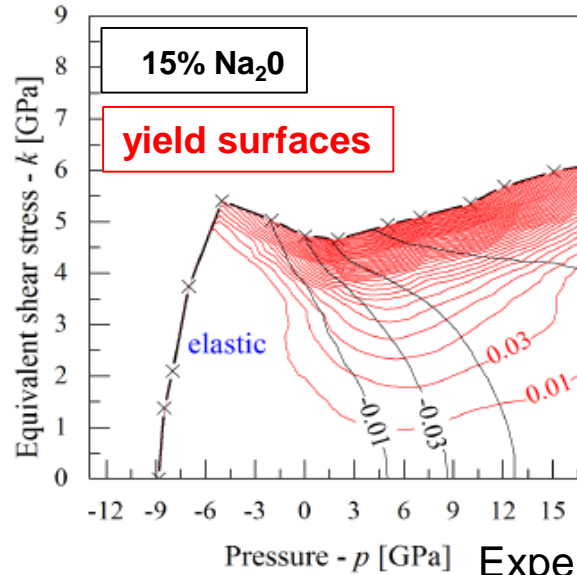
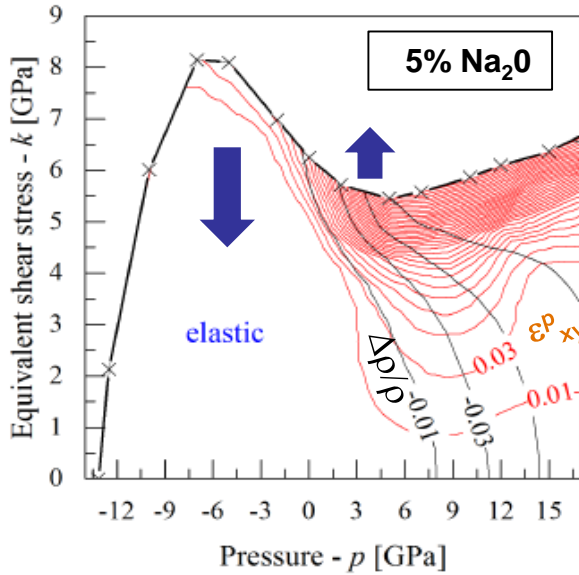
High loca Na+ density ↔ Low Elastic Modulus

G. Molnar et al. (2016)

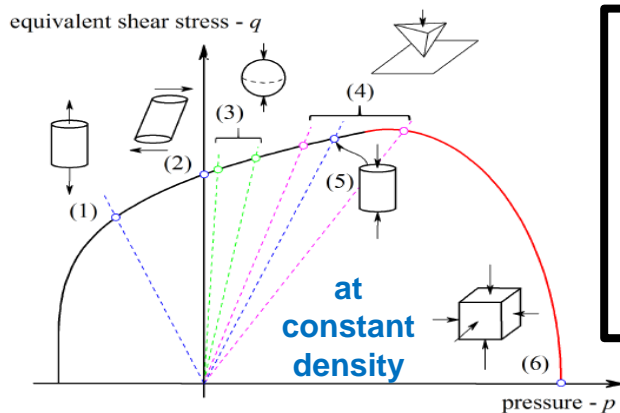
Sensitivity to Composition and Pressure in $(1-x) \text{SiO}_2 + x \text{Na}_2\text{O}$

Densification

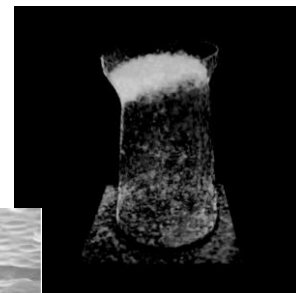
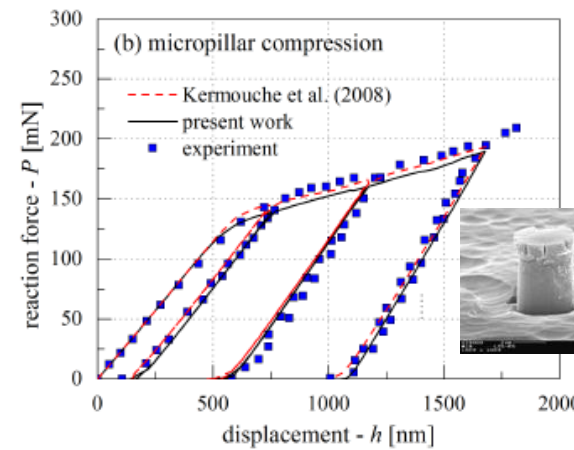
Shear as % Na₂O ↗



Experiments



Adapt the
parameters of
the yield
surfaces to
experimental
results



G. Kermouche.
(2022)



- I. Phenomenology
- II. Mechanical Instability
- III. Plastic Flow**



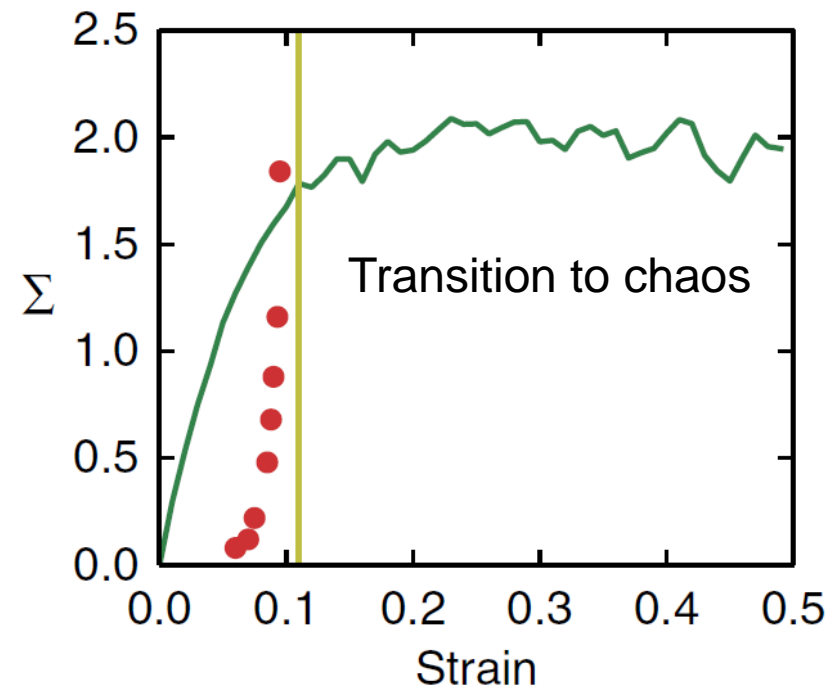
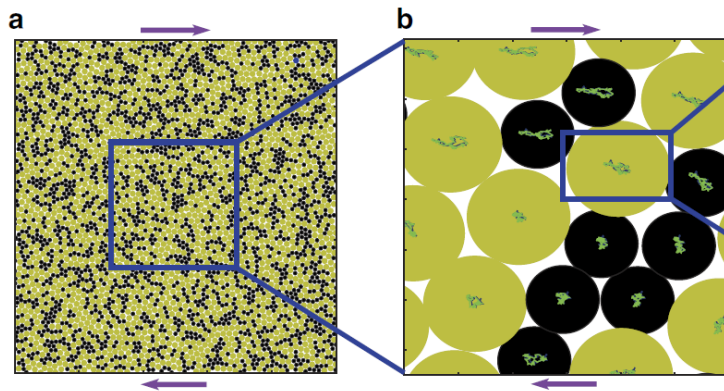
Received 31 Dec 2014 | Accepted 6 Oct 2015 | Published 13 Nov 2015

DOI: 10.1038/ncomms9805

OPEN

Reversibility and criticality in amorphous solids

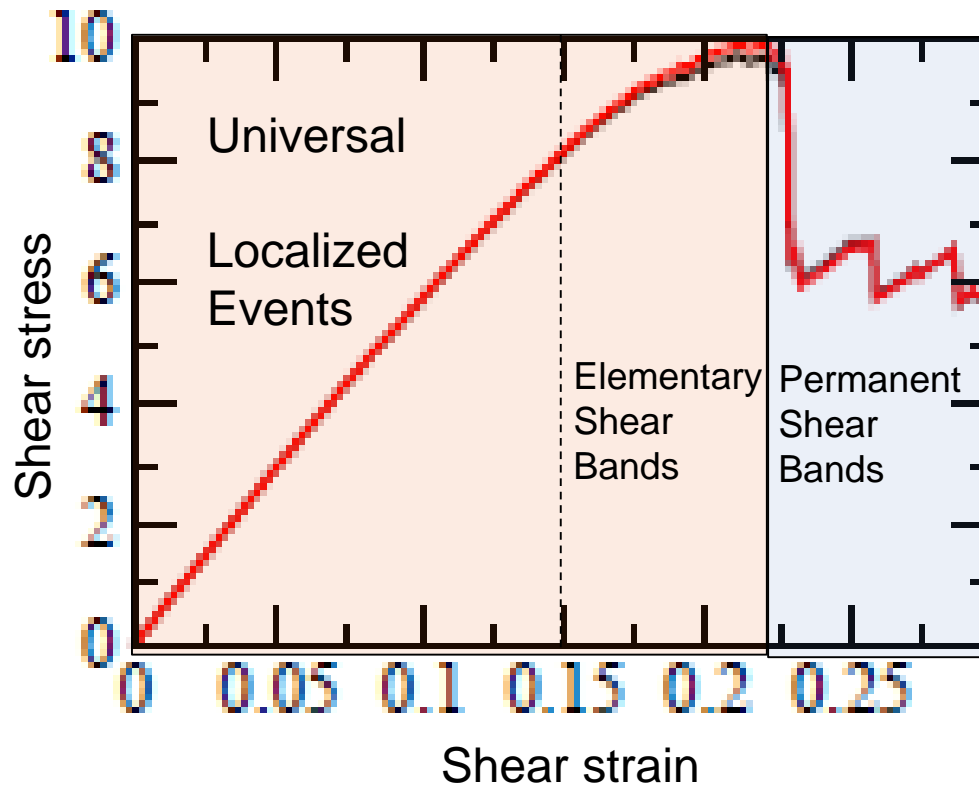
Ido Regev^{1,2,3}, John Weber⁴, Charles Reichhardt^{2,3}, Karin A. Dahmen⁴ & Turab Lookman^{2,3}





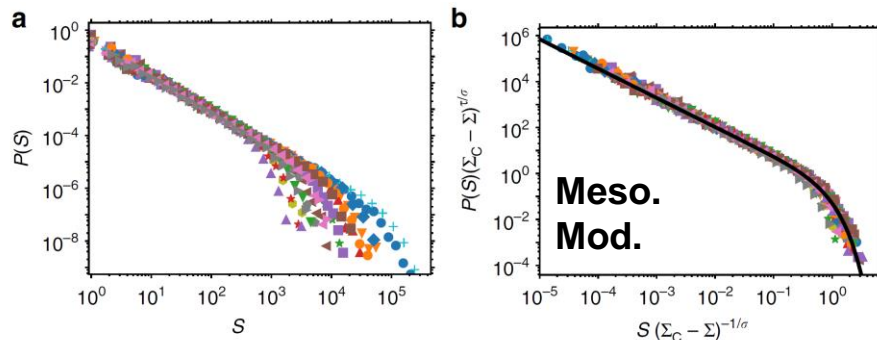
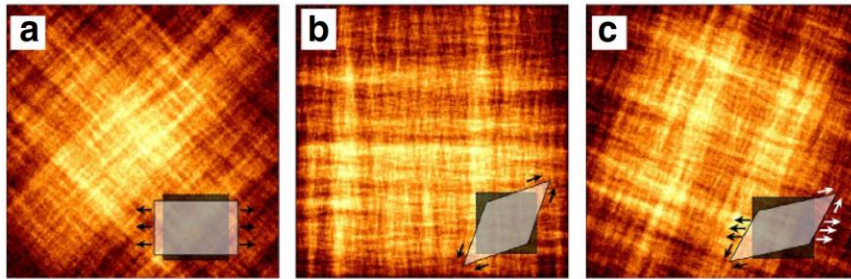
General Behaviour for Plastic Flow in Amorphous materials

Below Global Yield Stress ← → Above Global Yield Stress



Critical Behaviour below Yielding

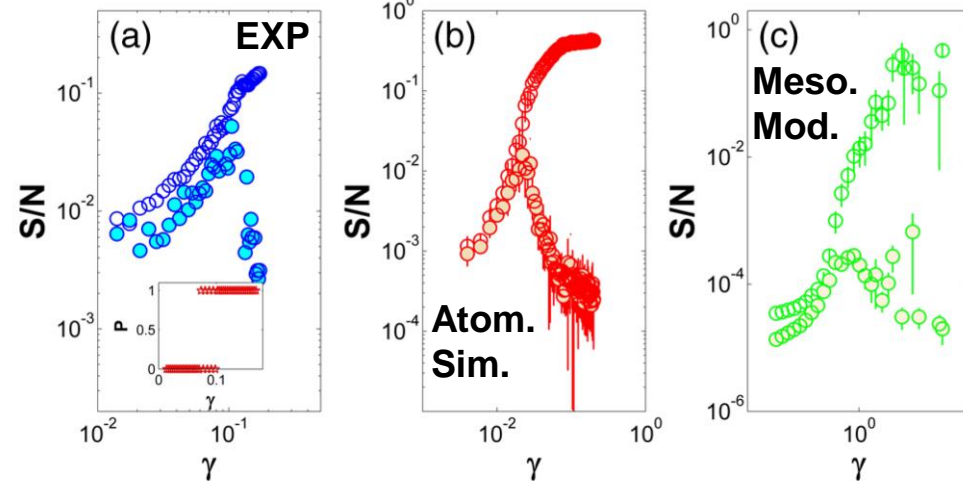
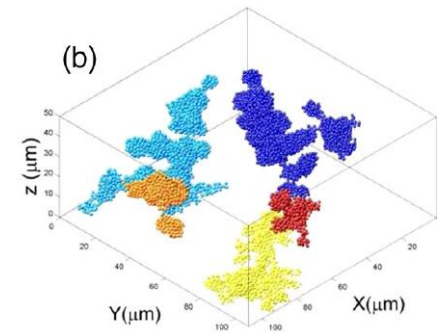
Avalanche Size distribution



Pure shear	Pure tension	Biaxial	Simple shear	Pure tension	3D pure shear	3D
● 0.995	▲ 0.994	▶ 0.966	+ 0.992	▼ 0.978	■ 0.982	
▽ 0.991	■ 0.989	● 0.961	◆ 0.986	★ 0.973	◀ 0.977	
★ 0.986	◀ 0.985	● 0.953	● 0.98	▲ 0.968	▶ 0.972	

Z. Budrikis et al. (2017)

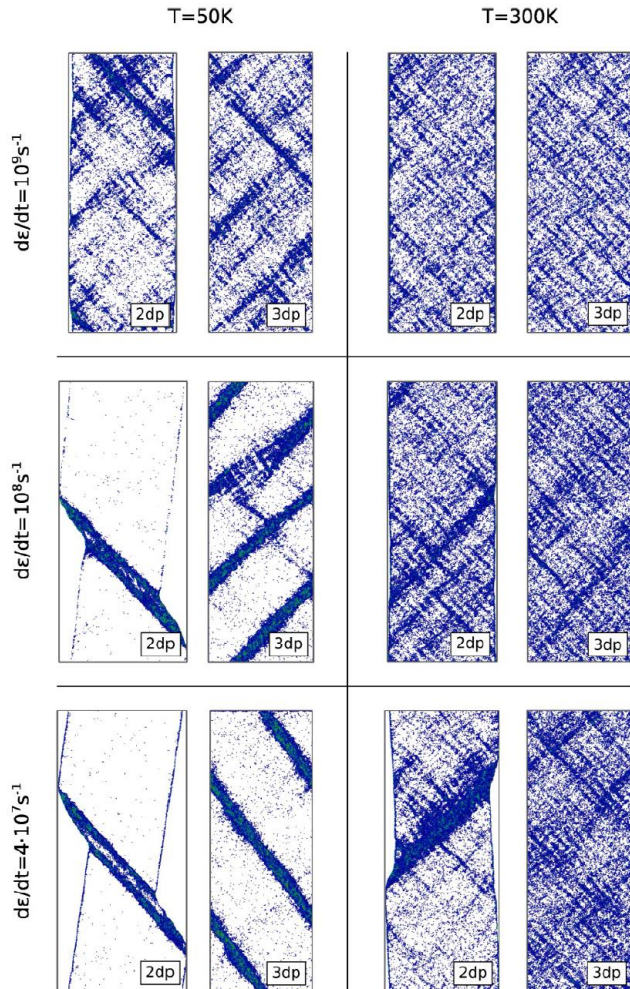
Percolation of Active Clusters



P. Schall et al. (2017)



The Occurrence of Permanent Shear-Banding in Non-Universal



Role of
Temperature

Strain Rate,
Free Boundaries

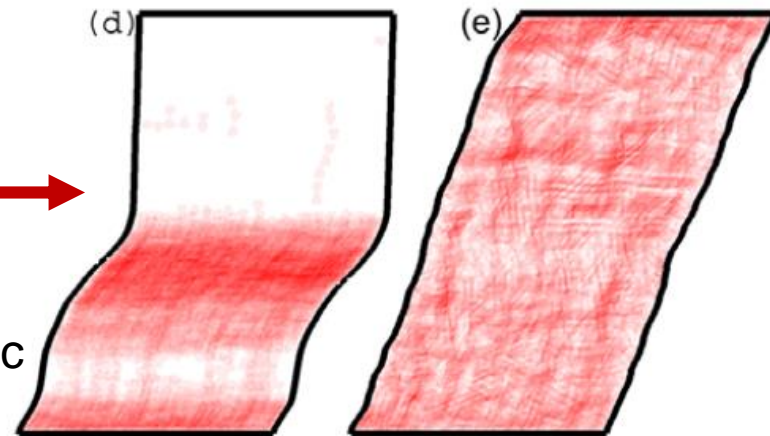


Importance of
Self-Softening

« Free Volume »
creation at Plastic
Rearrangement



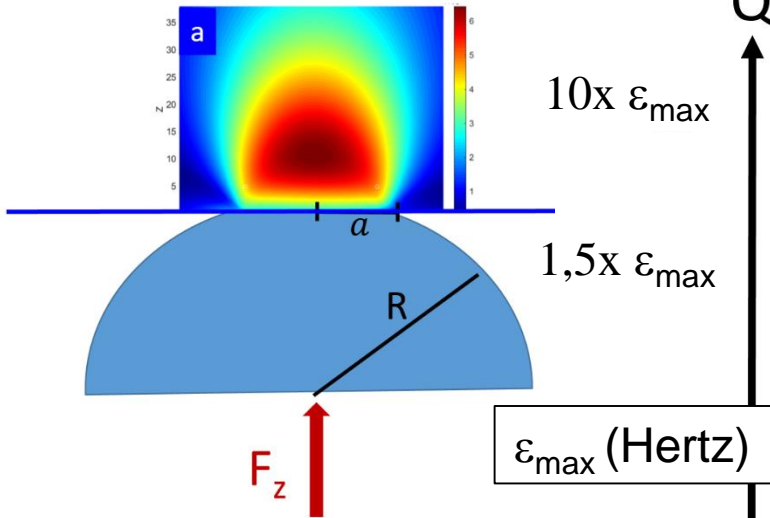
Non-universal
Process





Laboratoire de Mécanique des Contacts et des Structures

Example of shear banding in a contact problem

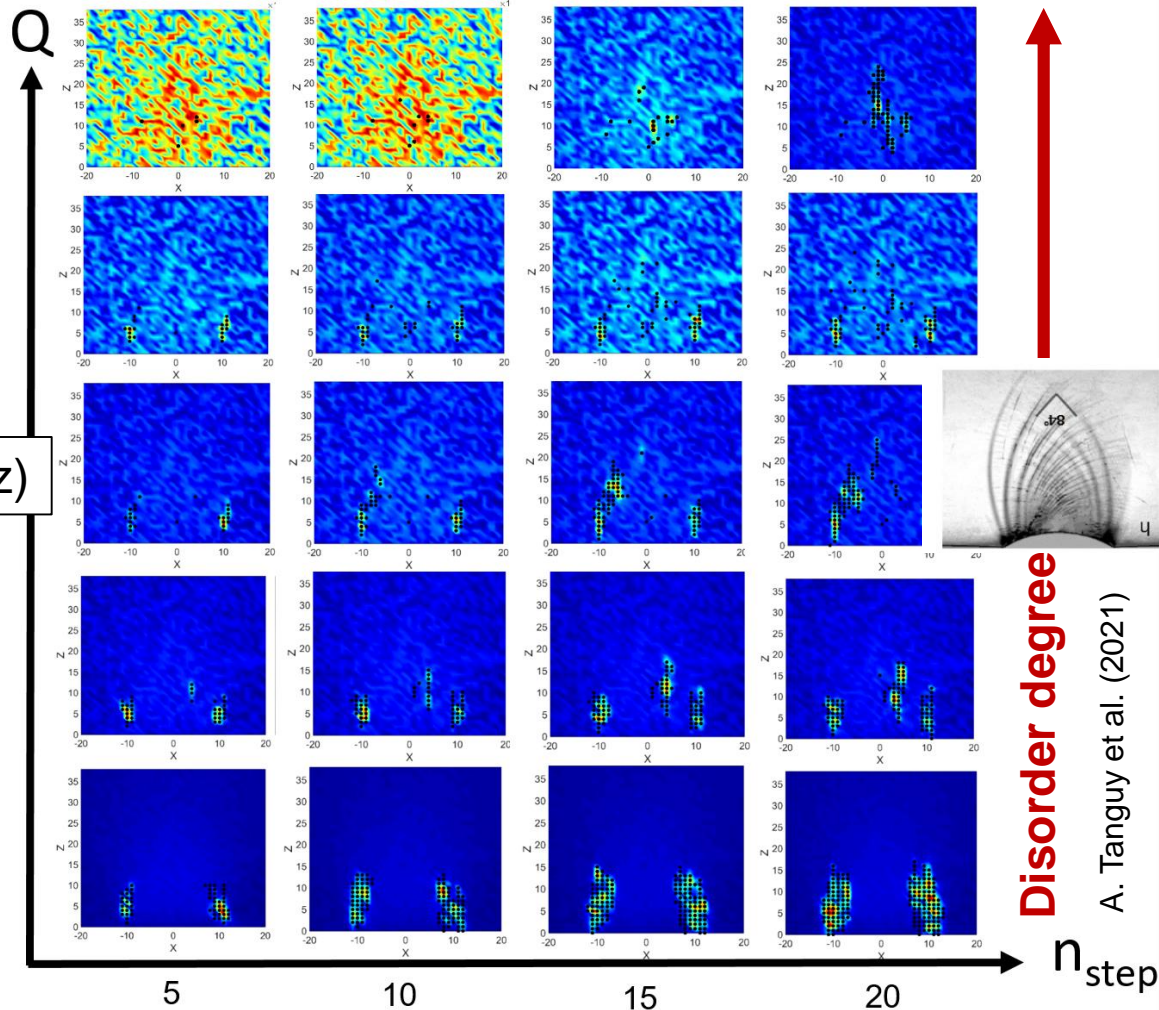


$\epsilon(i) > \epsilon_{\text{thresh}}^d(i)$: plastic event
then : $\epsilon(i) = \epsilon(i) + d\epsilon$ with stress redistribution

Random threshold:
 $\epsilon_{\text{thresh}}^d(i) = \epsilon_{\text{thresh}}^0 + Q \cdot \text{ran}(i)$

with initial surface defects

Sensitivity to Local Disorder

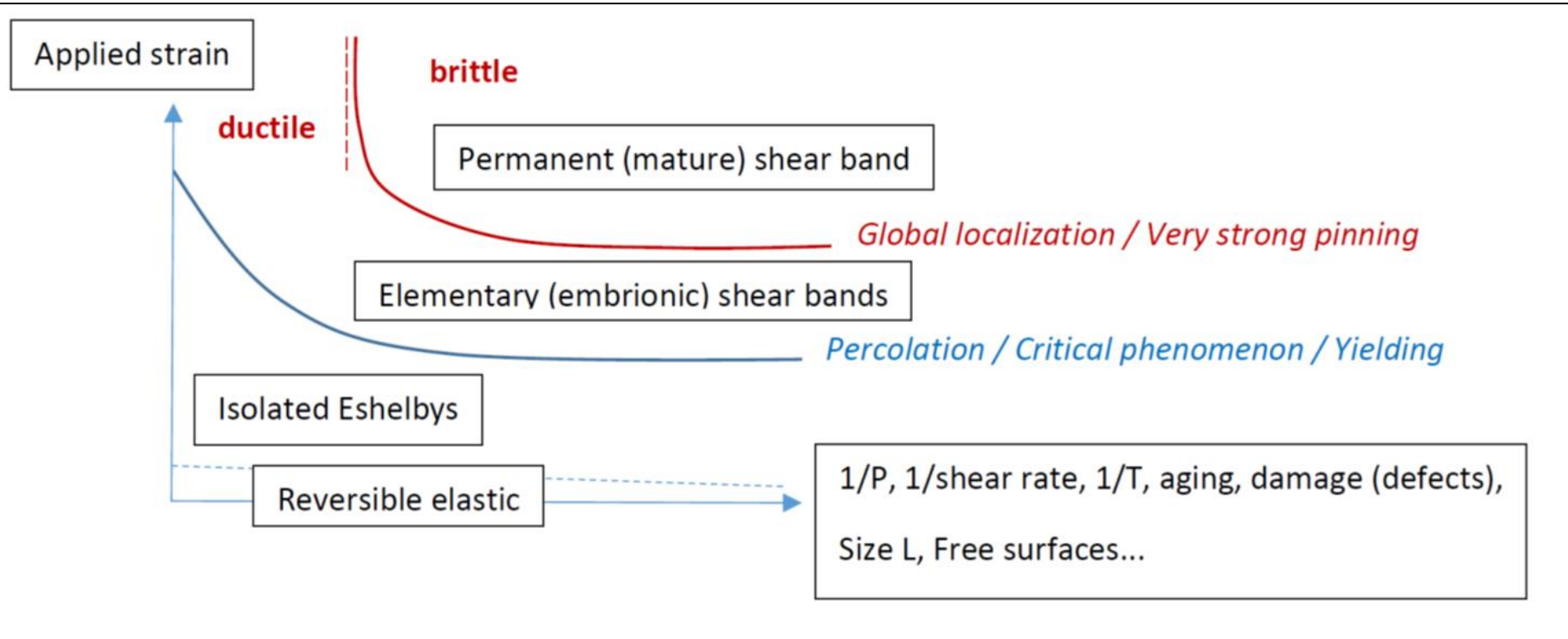


Disorder degree

A. Tanguy et al. (2021)



Summary





Conclusion

Universal features:

Plasticity as a Mechanical Instability (atomistic buckling)

Elementary events (Eshelby inclusions and Elementary – embrionic - shear bands)

Critical behaviour before yielding

Non-universal features have engineering importance:

Structural signature of plastic rearrangements

Number and spatial distribution of plastic events as a function of temperature,
as a function of composition, and as a function of strain rate

Importance of Self-softening or strong heterogeneities for permanent shear-banding

Sensitivity of the permanent – mature – shear-banding to the boundary conditions

Comparison to crystals:

Difficulty to identify a permanent structural signature of plasticity

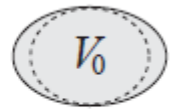
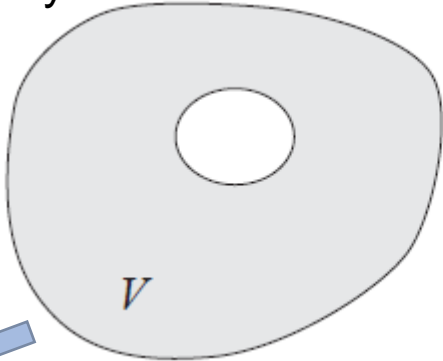
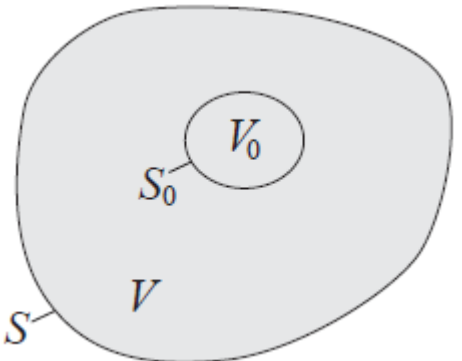
Coupling between shear and densification

No order to disorder transition

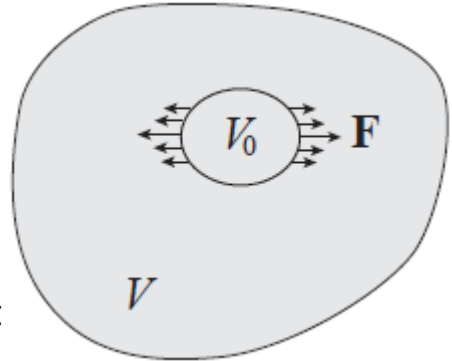
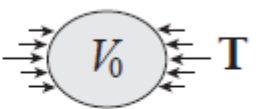
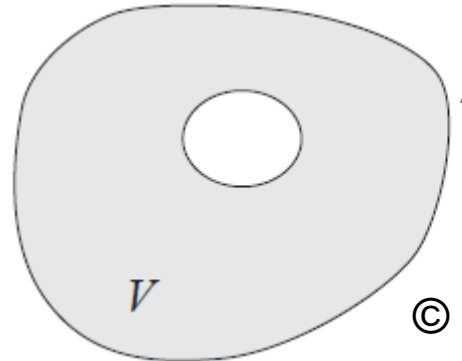


Eshelby Inclusion (1957)

Linear Elastic Heterogeneity
with ellipse shape
in a Linear Elastic matrix
 ε^* is the « eigenstrain »
or stress-free strain
Inside the inclusion



matrix	inclusion
$e_{ij} = 0$	$e_{ij} = e_{ij}^*$
$\sigma_{ij} = 0$	$\sigma_{ij} = 0$
$u_i = 0$	$u_i = e_{ij}^* x_j$



© C. Weinberger, W. Cai, D. Barnett

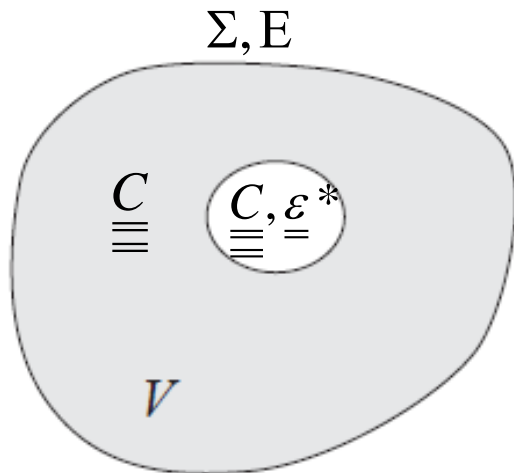
matrix	inclusion
$e_{ij} = 0$	$e_{ij} = e_{ij}^{el} + e_{ij}^* = 0$
$\sigma_{ij} = 0$	$\sigma_{ij} = C_{ijkl} e_{ij}^{el} = -C_{ijkl} e_{ij}^* = -\sigma_{ij}^*$
$u_i = 0$	$u_i = 0$

matrix	inclusion
$e_{ij} = e_{ij}^c$	$e_{ij} = e_{ij}^c$
$\sigma_{ij} = \sigma_{ij}^c$	$\sigma_{ij} = \sigma_{ij}^c - \sigma_{ij}^* = C_{ijkl}(e_{kl}^c - e_{kl}^*)$
$u_i = u_i^c$	$u_i = u_i^c$



Similar Cases:

1) Uniform Load at infinity:

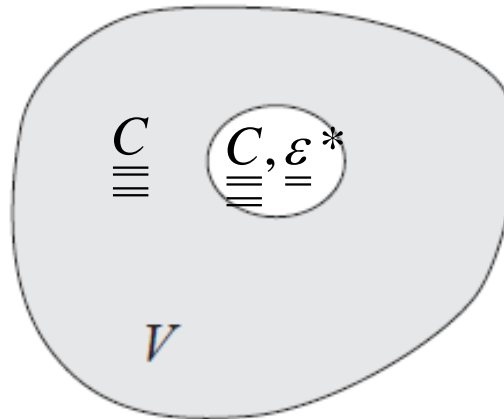


$$\underline{\underline{\sigma}}^I = \underline{\underline{\Sigma}} + \underline{\underline{C}} : \left(\underline{\underline{S}} - \underline{\underline{I}} \right) : \underline{\underline{\varepsilon}}^*$$

$$\underline{\underline{\varepsilon}} = \underline{\underline{E}} + \underline{\underline{S}} : \underline{\underline{\varepsilon}}^*$$

$$\underline{\underline{\sigma}}^M = \underline{\underline{\Sigma}} + \underline{\underline{C}} : \underline{\underline{S}} : \underline{\underline{\varepsilon}}^*$$

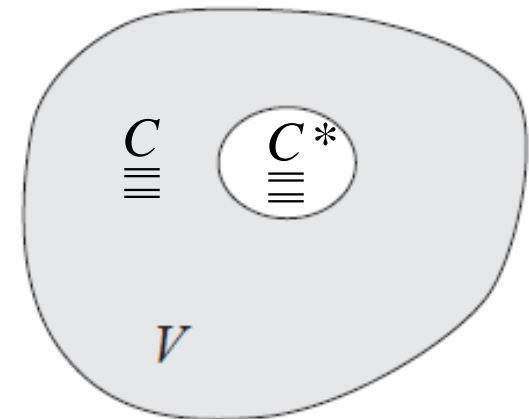
2) Empty Inclusion:



$$\underline{\underline{\Sigma}} + \underline{\underline{C}} : \left(\underline{\underline{S}} - \underline{\underline{I}} \right) : \underline{\underline{\varepsilon}}^* = \underline{\underline{0}}$$

$$\underline{\underline{\sigma}}^M = \underline{\underline{\Sigma}} + \underline{\underline{C}} : \underline{\underline{S}} : \underline{\underline{\varepsilon}}^*$$

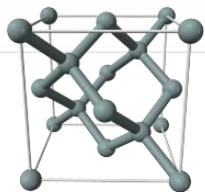
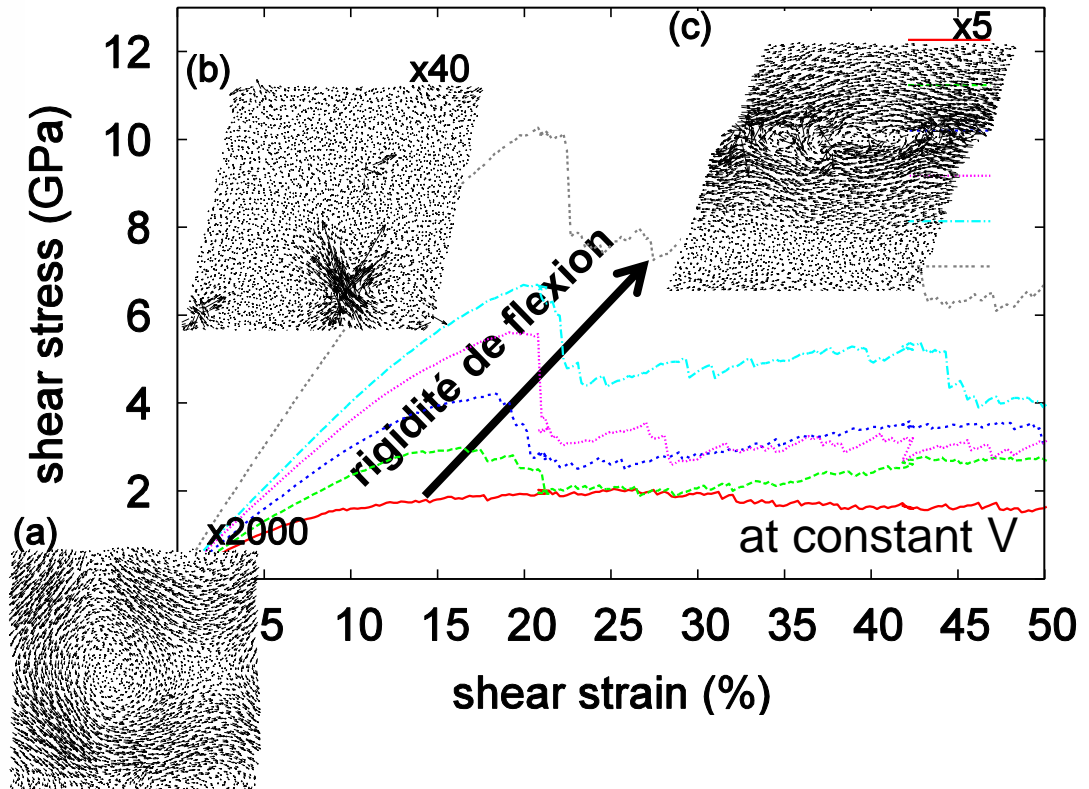
3) Elastic Inhomogeneity:



$$\underline{\underline{\sigma}}^I = \underline{\underline{\Sigma}} + \underline{\underline{C}} : \left(\underline{\underline{S}} - \underline{\underline{I}} \right) : \underline{\underline{\varepsilon}}^* = \underline{\underline{C^*}} : \underline{\underline{\varepsilon}}^I$$

$$\underline{\underline{\varepsilon}}^I = \underline{\underline{E}} + \underline{\underline{S}} : \underline{\underline{\varepsilon}}^*$$

The Example of Amorphous Silicon



$$E_{SW} = \sum_{(i,j)} f(r_{ij}) + \sum_{i,j,k} \Lambda \cdot \left(\cos \theta_{jik} + \frac{1}{3} \right)^2 \cdot e^{\gamma \cdot (r_{ij}-a)^{-1} + \gamma \cdot (r_{ik}-a)^{-1}}$$